



## Recent advances in application of transition metal phosphides for photocatalytic hydrogen production

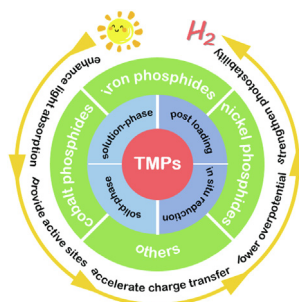
Yang Yang, Chengyun Zhou, Wenjun Wang, Weiping Xiong, Guangming Zeng\*, Danlian Huang\*, Chen Zhang\*, Biao Song, Wenjing Xue, Xiaopei Li, Ziwei Wang, Donghui He, Hanzhuo Luo, Zenglin Ouyang

College of Environmental Science and Engineering, Hunan University and Key Laboratory of Environmental Biology and Pollution Control (Hunan University), Ministry of Education, Changsha 410082, PR China

### HIGHLIGHTS

- The functions of TMPs in photocatalytic H<sub>2</sub> production are discussed.
- The synthetic strategies of TMPs and the loading methods of TMPs on semiconductors are introduced.
- The applications and mechanisms of the common TMPs in photocatalytic H<sub>2</sub> production are summarized.
- The challenges and opportunities of TMPs in photocatalytic H<sub>2</sub> production are proposed.

### GRAPHICAL ABSTRACT



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### ABSTRACT

Searching a sustainable way to efficiently produce hydrogen (H<sub>2</sub>) is critical to realizing the “hydrogen economy”, which may resolve the global energy and environmental issues nowadays. The conversion of solar energy to hydrogen energy based on photocatalytic water splitting is an ideal technology for environmental-friendly and economically producing H<sub>2</sub>. Exploring high-performance and earth-abundant cocatalysts that can replace noble metal-based cocatalysts is essential to achieving highly efficient and cost-effective photocatalytic H<sub>2</sub> production. In recent years, transition metal phosphides (TMPs) have been regarded as promising candidates to replace noble metal-based cocatalysts for photocatalytic H<sub>2</sub> production. This review presents a panorama of the latest progress in the developments of TMPs for photocatalytic H<sub>2</sub> production. Concretely, this review starts with the functions of TMPs in photocatalytic H<sub>2</sub> production, followed by the synthetic strategies of TMPs and the loading methods of TMPs on semiconductors. Then the application and mechanism of the common TMPs in photocatalytic H<sub>2</sub> production are discussed in detail, including iron phosphides, cobalt phosphides and nickel phosphides. Lastly, we provide a comprehensive conclusion and outlook on the major challenges and opportunities for better developments in the future research. It is reasonable to believe that the TMPs is a rising star in photocatalytic H<sub>2</sub> production.

\* Corresponding authors.

E-mail addresses: [zgming@hnu.edu.cn](mailto:zgming@hnu.edu.cn) (G. Zeng), [huangdanlian@hnu.edu.cn](mailto:huangdanlian@hnu.edu.cn) (D. Huang), [zhangchen@hnu.edu.cn](mailto:zhangchen@hnu.edu.cn) (C. Zhang).

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## 1. Introduction

The growing consumption of energy and the consequent climate changes as well as environmental problems make it urgent to seek renewable and environment-friendly alternative energy sources to fossil fuels [1–5].  $H_2$  is considered as the most development potential candidate for the future energy supply owing to its high gravimetric energy density and calorific value, as well as clean combustion product ( $H_2O$ ) [6–8]. At present, the main industrial method is  $H_2$  production from fossil resources, including gasification of coal, steam reforming of natural gas and partial oxidation of hydrocarbons [9–11]. Although a considerable amount of  $H_2$  can be produced at low cost in these processes, the consumption of fossil resources will lead to the greenhouse gas emission. Thus, developing a green, sustainable and efficient strategy for  $H_2$  production is imperatively required.

Since Fujishima et al. [12] developed the photoelectrochemical water splitting on a rutile  $TiO_2$  anode in 1972, photocatalytic water splitting using semiconductor photocatalysts has become a promising technology to convert solar energy into hydrogen energy [13,14]. This is an environmental-friendly technology as  $H_2$  can be produced through the interaction among solar light, photocatalyst and water in this process. Moreover, the easy synthesis of photocatalysts and the facile design of photoreactors make photocatalytic water splitting a low cost technology with the potential for large scale  $H_2$  production [15,16]. A technical/economic analysis has demonstrated that when some photoreactors were operated with a solar-to-hydrogen (STH) energy conversion efficiency of 5–10%, photocatalytic water splitting was economically feasible for the  $H_2$  production [17,18]. As displayed in Fig. 1a, the number of publications related to photocatalytic  $H_2$  production shows an exponential growth trend. Generally, the photocatalytic  $H_2$  production reaction includes three main steps: (i) semiconductor absorbs light to generate electron-hole pairs, (ii) photogenerated electron-hole pairs are separated and transferred to the surface of semiconductor, and (iii)  $H_2$  evolution on the surface of semiconductor (Fig. 2). The relative balance of kinetics and thermodynamics of these three steps influences the efficiency of photocatalytic  $H_2$  production. Therefore, many efforts have been paid to construct novel photocatalysts with broad absorption spectrum and efficient charge separation and transfer in the last few decades [19–21]. For example, some novel photocatalysts with narrow bandgap have been developed, such as  $Ag_3PO_4$ ,  $Bi_2WO_6$  and  $g-C_3N_4$  [22–25]. Meanwhile, in order to improve the light harvesting ability of wide bandgap photocatalysts, a variety of strategies have also been exploited, such as heterojunction construction [26,27], plasmonic metals coupling [28], dye sensitization [29], doping [30] and defect engineering [31]. In addition, various nanostructured photocatalysts have been designed to shorten charge diffusion length, thus achieving highly efficient charge

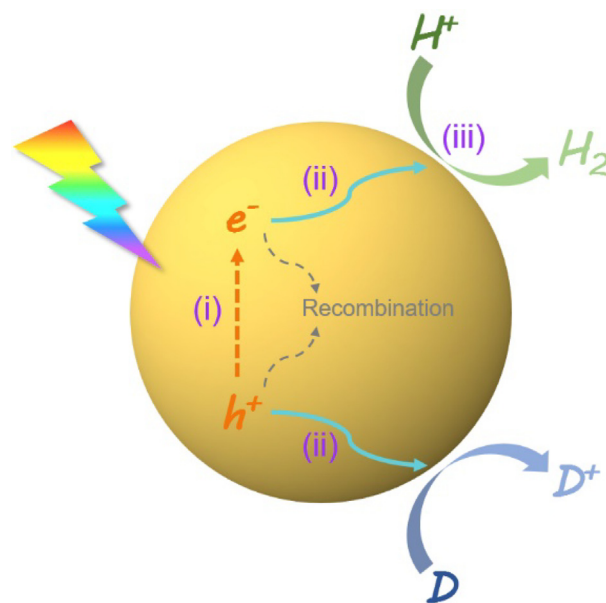


Fig. 2. Schematic illustration of photocatalytic  $H_2$  production over a semiconductor photocatalyst.

separation and transfer [32]. Moreover, the heterojunction construction [33,34], doping [35,36] and defect engineering [37] are also widely employed to accelerate the charge separation and transfer.

In the last decades, cocatalysts loading has stimulated much attention as it can not only enhance the light harvesting and assist charge transfer but also provide reaction active sites and improve photocatalysts stability [38,39]. Thus, cocatalysts play an important role for enhancing both the activity and stability of photocatalysts [40–42]. Among the various cocatalysts, noble metal-based cocatalysts have been extensively studied as highly active cocatalysts toward photocatalytic  $H_2$  production, such as Pt, Pd, Au, Ag, Ru and Rh [43,44]. Nevertheless, the scarcity and accompanying expensive price limit the large scale application of these noble metal-based cocatalysts (Table 1). Accordingly, the development of noble metal free cocatalysts with low cost and high efficiency is urgently demanded for photocatalytic  $H_2$  production. Recently, earth-abundant transition metal phosphides (TMPs) have been recognized as affordable and efficient catalysts for hydrogen evolution reaction (HER) and have drawn increasing attention (Fig. 1b). As early as in 2005, Liu et al. [45] predicted that  $Ni_2P$  was a highly active HER catalyst using density functional theory (DFT) calculations. In 2013, Xu et al. [46] first reported that FeP had excellent activity in electrochemical  $H_2$  evolution. Since Cao et al. [47] first

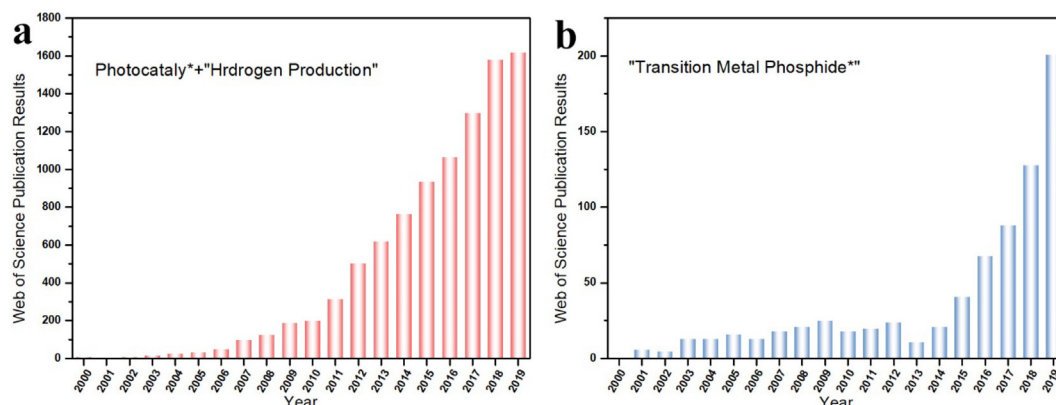


Fig. 1. Histogram of number of publications on (a) photocatalytic  $H_2$  production and (b) transition metal phosphides from 2000 to 2019 (date source: Web of Science).

**Table 1**  
Price of common cocatalysts.

Cocatalysts	CAS Number	Properties	Price (USD g <sup>-1</sup> )
Pt	7440-06-4	wire, diam. 2.0 mm, 99.9% trace metals basis	376.43
Pd	7440-05-3	powder, < 1 μm, ≥ 99.9% trace metals basis	153.00
Au	7440-57-5	nanopowder, < 100 nm particle size, 99.9% trace metals basis	405.00
Ag	7440-22-4	nanopowder, < 150 nm particle size, 99% trace metals basis	31.20
Ru	7440-18-8	powder, 99.99% trace metals basis	211.00
Rh	7440-16-6	powder, 99.95% trace metals basis	611.00
Fe <sub>2</sub> P	1310-43-6	99.5% trace metals basis	18.18
Ni <sub>2</sub> P	12035-64-2	– 100 mesh, 98%	11.40

CAS: Chemical Abstracts Service; USD: United States dollar. (Price data came from Sigma-Aldrich: 7/20/2020).

demonstrated that Ni<sub>2</sub>P was an outstanding cocatalyst for photocatalytic H<sub>2</sub> production in 2014, many efforts have been devoted to introducing TMPs into photocatalytic H<sub>2</sub> production system [48–52].

Because many achievements have been made in developing TMPs cocatalysts for photocatalytic H<sub>2</sub> production in recent years, it is necessary to summarize the recent progresses and thus promote further developments of this field. In this review, we summarize the recent advances of TMPs in photocatalytic H<sub>2</sub> production. Firstly, the functions of TMPs in photocatalytic H<sub>2</sub> production are discussed. Then, the synthetic strategies of TMPs and the loading methods of TMPs on semiconductors are introduced. Next, the applications and mechanisms of the common TMPs in photocatalytic H<sub>2</sub> production are stated. Finally, the challenges and opportunities of TMPs in photocatalytic H<sub>2</sub> production are proposed.

## 2. Functions of TMPs in photocatalytic H<sub>2</sub> production

In photocatalytic H<sub>2</sub> production, the TMPs cocatalysts play significant roles in promoting both the activity and stability of semiconductor photocatalysts, mainly including enhancing light absorption, providing active sites, accelerating charge transfer, lowering overpotential and strengthening photostability.

### 2.1. Enhance light absorption

Light harvesting capability is an important factor influencing photocatalytic activity because the enhanced light absorption ability can improve the utilization efficiency of solar energy [53]. TMPs have been reported to effectively enhance the light absorption due to their dark appearance. For example, in the study of Zhao et al. [54], the pure Fe<sub>x</sub>P presented an intense absorption in the wavelength range from 350 to 800 nm, while the absorption edge of pure g-C<sub>3</sub>N<sub>4</sub> located at approximately 450 nm. After introducing Fe<sub>x</sub>P into g-C<sub>3</sub>N<sub>4</sub>, although the absorption edge of Fe<sub>x</sub>P/g-C<sub>3</sub>N<sub>4</sub> did not change significantly compared to g-C<sub>3</sub>N<sub>4</sub>, an enhanced light absorption was observed in Fe<sub>x</sub>P/g-C<sub>3</sub>N<sub>4</sub> in the wavelength range from 450 to 800 nm, which was ascribed to the high visible light absorption ability of Fe<sub>x</sub>P. Huang et al. [55] found that the pristine CoP<sub>3</sub> possessed an efficient absorption from the visible light to near-infrared light region because of its narrow band gap. Benefiting from the intrinsic absorption of black-colored CoP<sub>3</sub>, the CoP<sub>3</sub>/Mn<sub>0.2</sub>Cd<sub>0.8</sub>S nanocomposite exhibited enhanced light absorption. Moreover, the Ni<sub>2</sub>P, Ni<sub>12</sub>P<sub>5</sub> and Ni<sub>3</sub>P could also efficiently enhance the light absorption of g-C<sub>3</sub>N<sub>4</sub> because of their dark color [56]. Accordingly, the TMPs cocatalysts can significantly enhance the light absorption, which will promote the generation of more available photo-generated electrons to participate in the photocatalytic H<sub>2</sub> production.

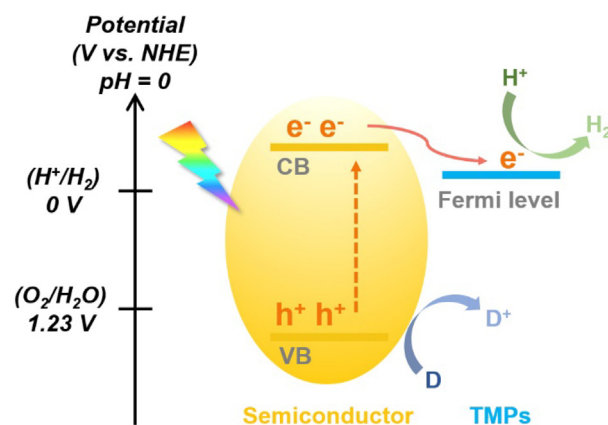
### 2.2. Provide active sites

Constructing surface active sites to achieve rapid charge transfer and effective water molecule adsorption is beneficial to improve the performance of the photocatalyst for photocatalytic H<sub>2</sub> production

[57]. TMPs can provide abundant active sites and push forward photocatalytic H<sub>2</sub> production. For example, in the CoP/g-C<sub>3</sub>N<sub>4</sub> composite, a large number of photogenerated electrons transferred from g-C<sub>3</sub>N<sub>4</sub> to the surface CoP catalytic active sites under visible light irradiation [58]. Then the enriched electrons on CoP reacted with adsorbed protons to produce H<sub>2</sub>. Thus, CoP could act as the electron sink and provide effective proton reduction sites to promote H<sub>2</sub> production. Moreover, both metal atom and P atom in TMPs can be active sites. Li et al. [59] revealed that the P(δ<sup>-</sup>)-Co(δ<sup>+</sup>)-N(δ<sup>-</sup>) bonding state was formed among P, Co, and N atoms in tertiary nitrogen groups on the interface of CoP and g-C<sub>3</sub>N<sub>4</sub>. The Co(δ<sup>+</sup>)-N(δ<sup>-</sup>) bonds could stabilize the CoP cocatalyst on the surface of g-C<sub>3</sub>N<sub>4</sub>, promote the photogenerated charge carrier separation and transfer, and regulate the CoP activity. Meanwhile, the P(δ<sup>-</sup>)-Co(δ<sup>+</sup>) bonds were favorable to the formation of high electron density protonated P(δ<sup>-</sup>) pendant moieties and Co-hydride, which made the adjacent P and Co atoms serve as dual proton adsorption sites to accelerate the photocatalytic H<sub>2</sub> production from water. Briefly, the TMPs cocatalysts can efficiently increase the surface active sites to promote the photocatalytic H<sub>2</sub> production.

### 2.3. Accelerate charge transfer

The transfer behavior of photogenerated charge carriers is a significant factor affecting the photocatalytic activity. TMPs cocatalysts can accelerate interfacial charge transfer at the interface of TMPs and semiconductors due to their intrinsic metallicity [60,61]. When TMPs is loaded on the surface of semiconductors, a close contact can be established between their interfaces, which leads to the formation of heterojunction, hence accelerating the separation and transfer of photo-generated electron-hole pairs. As depicted in Fig. 3, the photogenerated electrons on the conduction band (CB) of a semiconductor will migrate to the TMPs and reduce protons to H<sub>2</sub> molecules. In general, the relative energy level between semiconductor and TMPs determines the direction and efficiency of the charge transfer [43]. In the study of Wang



**Fig. 3.** Description of band diagram for photocatalytic H<sub>2</sub> production over a semiconductor photocatalyst loaded with TMPs cocatalyst.

et al. [62], the Fermi level of CoP was determined to be lower than that of g-C<sub>3</sub>N<sub>4</sub>. When CoP was loaded on the surface of g-C<sub>3</sub>N<sub>4</sub>, the free electrons would transfer from g-C<sub>3</sub>N<sub>4</sub> to CoP until the Fermi levels were aligned, which would induce the formation of built-in electric field oriented from g-C<sub>3</sub>N<sub>4</sub> towards CoP. Therefore, under visible light irradiation, the photogenerated electrons on g-C<sub>3</sub>N<sub>4</sub> would be directionally transferred to CoP to produce H<sub>2</sub>. He et al. [63] discovered that the Ni<sub>2</sub>P could be used as an interface electronic bridge to link the two conduction bands of g-C<sub>3</sub>N<sub>4</sub> and CdS for facilitating the transfer of photogenerated charge carriers. DFT calculations demonstrated that the work functions of g-C<sub>3</sub>N<sub>4</sub>, Ni<sub>2</sub>P and CdS were 4.23, 4.52 and 5.18 eV, respectively. Accordingly, the Fermi level was g-C<sub>3</sub>N<sub>4</sub>, Ni<sub>2</sub>P and CdS in descending order. During photocatalytic H<sub>2</sub> production, the photogenerated electrons migrated from g-C<sub>3</sub>N<sub>4</sub> to Ni<sub>2</sub>P and CdS or migrated from Ni<sub>2</sub>P to CdS, resulting in the formation of built-in electric field, thereby efficiently promoting the separation and transfer of photogenerated charge carriers. Compared to CdS/g-C<sub>3</sub>N<sub>4</sub>, the photoluminescence (PL) intensity of CdS/Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> was decreased, implying the positive role of Ni<sub>2</sub>P on the charge carrier dynamics. Moreover, Lin et al. [64] reported that the appropriate charge transfer distance between the semiconductor and TMPs was also a key factor for accelerating charge transfer. In short, the TMPs cocatalysts can efficiently accelerate charge transfer and enable more photogenerated electrons to get involved in the process of H<sub>2</sub> production.

#### 2.4. Lower overpotential

Overpotential is the difference between the thermodynamically determined reduction potential and the experimental potential, which is mainly used to overcome the inherent activation energy barrier during electrode reaction and the resistance consumption [50,65]. However, in the process of photocatalytic water splitting, semiconductors always have a high surface overpotential for HER, resulting in the decrease of reaction rate and the loss of energy [66,67]. It has been found that TMPs cocatalysts could efficiently lower the overpotential for HER on the surface of semiconductors. In the study of Zhen et al. [68], the overpotential of CdS, Ni<sub>2</sub>P, Pt@CdS and Ni<sub>2</sub>P@CdS were determined to be -0.53, -0.38, -0.34 and -0.32 V vs. saturated calomel electrode (SCE), respectively. Among these, the Ni<sub>2</sub>P@CdS possessed the lowest overpotential, which might be ascribed to the fast transfer of electrons. Accordingly, the Ni<sub>2</sub>P@CdS exhibited excellent photocatalytic performance for H<sub>2</sub> production due to the low overpotential for HER. Sun et al. [69] reported that the Fe<sub>2</sub>P, Co<sub>2</sub>P and Ni<sub>2</sub>P could all cause a negative shift of onset potential for g-C<sub>3</sub>N<sub>4</sub>. The negative shift of onset potential indicated the reduced overpotential, which was beneficial to the photocatalytic H<sub>2</sub> production. Normally, the photocatalyst with lower overpotential requires less energy to obtain the same photocurrent density. In the study of Bi et al. [70], when the photocurrent density was -10 mA cm<sup>-2</sup>, the overpotential of NiCoP/g-C<sub>3</sub>N<sub>4</sub> was -1.66 V vs. Ag/AgCl, which was lower than that of the pristine g-C<sub>3</sub>N<sub>4</sub> (-1.88 V vs. Ag/AgCl). Meanwhile, a photocurrent density of 16.05 mA cm<sup>-2</sup> could be achieved on NiCoP/g-C<sub>3</sub>N<sub>4</sub> when potential was -1.8 V vs. Ag/AgCl, which was 2.3 times higher than that of the pristine g-C<sub>3</sub>N<sub>4</sub> (6.82 mA cm<sup>-2</sup>). The reduced overpotential and increased photocurrent density would be efficiently promote the photocatalytic activity for H<sub>2</sub> production. Moreover, it has been reported that the overpotential of Ni<sub>x</sub>P<sub>y</sub> was decreased with the decrease of ratio between Ni and P [71]. In the study of Sun et al. [56], Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> exhibited the lowest overpotential, similar to Pt/g-C<sub>3</sub>N<sub>4</sub>, followed by Ni<sub>12</sub>P<sub>5</sub>/g-C<sub>3</sub>N<sub>4</sub>, Ni<sub>3</sub>P/g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub> in sequence. In a word, the TMPs cocatalysts can significantly lower the overpotential of photocatalysts for HER, thereby improving the performance of photocatalytic H<sub>2</sub> production.

#### 2.5. Strengthen photostability

From the view of practical applications, the photostability of photocatalysts is critical in photocatalytic H<sub>2</sub> production. As displayed in Fig. 3, the loading of TMPs on a semiconductor can efficiently strengthen photostability of the semiconductor photocatalyst by transferring the photogenerated charge carriers in time [72,73]. For example, CdS is easily oxidized by photogenerated holes accumulated on its surface, leading to its self-decomposition [74]. Zhen et al. [68] developed core-shell structured Ni<sub>2</sub>P@CdS photocatalyst with high activity and stability for H<sub>2</sub> production under visible light irradiation. The Ni<sub>2</sub>P shell could restrain the self-decomposition of CdS core by separating the photogenerated holes for O<sub>2</sub> evolution. The Ni<sub>2</sub>P@CdS exhibited excellent activity after four cycles, while the pure CdS displayed low stability after four cycles, indicating that the Ni<sub>2</sub>P could suppress photocorrosion and improve stability of CdS. Meanwhile, the low concentration of Cd<sup>2+</sup> and S<sup>2-</sup> on the Ni<sub>2</sub>P@CdS reaction systems during the four cycles also testified the anti-photocorrosion role of Ni<sub>2</sub>P on CdS in the photocatalytic H<sub>2</sub> production. In the study of Cheng et al. [75], the FeP/CdS photocatalyst exhibited outstanding photostability in H<sub>2</sub> production even after 100 h of visible light irradiation. The strengthened photostability of FeP/CdS photocatalyst was attributed to the band bending between FeP and CdS, which accelerated the separation and transfer of photogenerated electron-hole pairs. Besides, black phosphorus (BP) also shows low stability in photocatalytic H<sub>2</sub> production because its edges are easily oxidized by dissolved oxygen [76]. Yuan et al. [77] found that the Co<sub>2</sub>P cocatalyst selectively grown on the edges of BP nanosheets could suppress the oxidation of BP nanosheets via a contact inhibitor, hence improving the stability of Co<sub>2</sub>P/BP photocatalyst. The H<sub>2</sub> production rate presented no obvious change and still remained at 97% after 12 h of photocatalytic reaction. Moreover, the high-resolution P 2p and Co 2p spectra before and after the reaction did not change significantly, further demonstrating the photostability of Co<sub>2</sub>P/BP photocatalyst during the photocatalytic H<sub>2</sub> production. In brief, the TMPs cocatalysts can efficiently strengthen the photostability of the photocatalysts in photocatalytic H<sub>2</sub> production by reducing the contact of semiconductor photocatalysts with photogenerated holes or dissolved oxygen.

### 3. Synthetic strategies

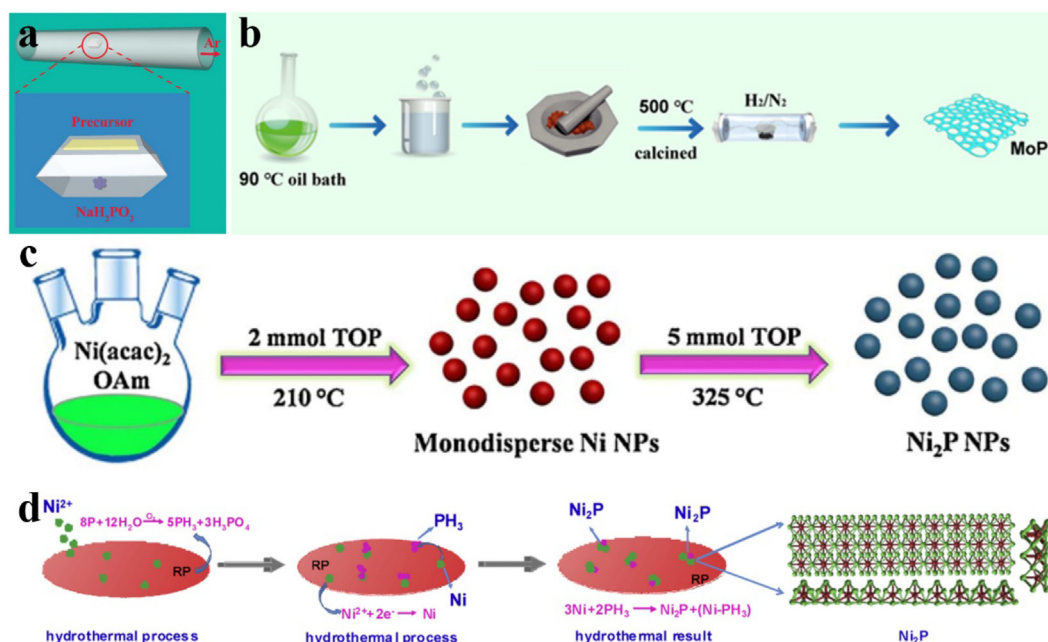
Based on the phase of reactants, there are mainly two strategies to prepare TMPs for photocatalytic H<sub>2</sub> production, including solid-phase synthetic method and solution-phase synthetic method.

#### 3.1. Solid-phase synthetic method

Solid-phase synthetic method is a process of mixing phosphorus source and solid metal source followed with thermal treatment under specific atmosphere. In this method, NaH<sub>2</sub>PO<sub>2</sub> is widely utilized as phosphorus source because it can release PH<sub>3</sub> when the temperature exceeds 250 °C. The generated PH<sub>3</sub> can further directly react with solid precursors such as metal oxides, metal hydroxides, and metal-organic frameworks (MOFs) to form TMPs, as displayed in Fig. 4a. For example, CoP nanoparticles was successfully prepared by calcining the mixture of Co<sub>3</sub>O<sub>4</sub> and NaH<sub>2</sub>PO<sub>2</sub> at 300 °C under N<sub>2</sub> atmosphere [78]. Li et al. [79] calcined the mixture of NiCo-MOF and NaH<sub>2</sub>PO<sub>2</sub> at 350 °C under Ar atmosphere to obtain NiCoP<sub>2</sub>. This method is beneficial to maintain the morphology and size of the precursors, but it is necessary to pay attention to the treatment of the toxic tail gas containing PH<sub>3</sub> in this process.

Moreover, direct reduction of metal orthophosphates by H<sub>2</sub> at a higher temperature is also a kind of solid-phase synthetic method to prepare TMPs, mainly utilized in MoP and WP. For example, Liu al. [80] acquired a MoP precursor by evaporating and drying a homogeneous solution containing (NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub>·4H<sub>2</sub>O, (NH<sub>4</sub>)<sub>2</sub>HPO<sub>4</sub>, citric





**Fig. 4.** (a) Schematic illustration of thermal phosphorization reaction. Reproduced with permission from Ref. [78] Copyright 2019 Royal Society of Chemistry. (b) Schematic synthesis of MoP. Reproduced with permission from Ref. [80] Copyright 2018 Elsevier. (c) Schematic synthesis of  $\text{Ni}_2\text{P}$  nanoparticles. Reproduced with permission from Ref. [82] Copyright 2018 Elsevier. (d) The possible formation mechanism process of  $\text{Ni}_2\text{P}$ /RP the composite. Reproduced with permission from Ref. [85] Copyright 2019 Elsevier.

acid and deionized water, and then calcining at 500 °C. Subsequently, they milled the MoP precursor and heated it at 850 °C under  $\text{H}_2/\text{N}_2$  atmosphere to get MoP (Fig. 4b). Unfortunately, the TMPs obtained by this method present a large size and irregular morphology, which may reduce the number of active sites. Besides, TMPs can also be synthesized by high energy ball milling method. In this process, a mixture of  $\text{NaH}_2\text{PO}_2$  or phosphorus powders and metal source will be subjected to high-energy collision from milling media. Then, the TMPs is formed through an annealing treatment in an inert atmosphere. For example, Hu et al. [81] prepared CoP by high energy ball milling the mixture of red phosphorus and cobalt powder and annealing at 200 °C. Although solid-phase synthetic method is a cost-effective strategy for large-scale synthesis of TMPs, it usually requires longer reaction times, higher reaction temperatures and protection of an inert atmosphere. Meanwhile, it is often difficult to precisely control the morphology, size and purity of the TMPs.

### 3.2. Solution-phase synthetic method

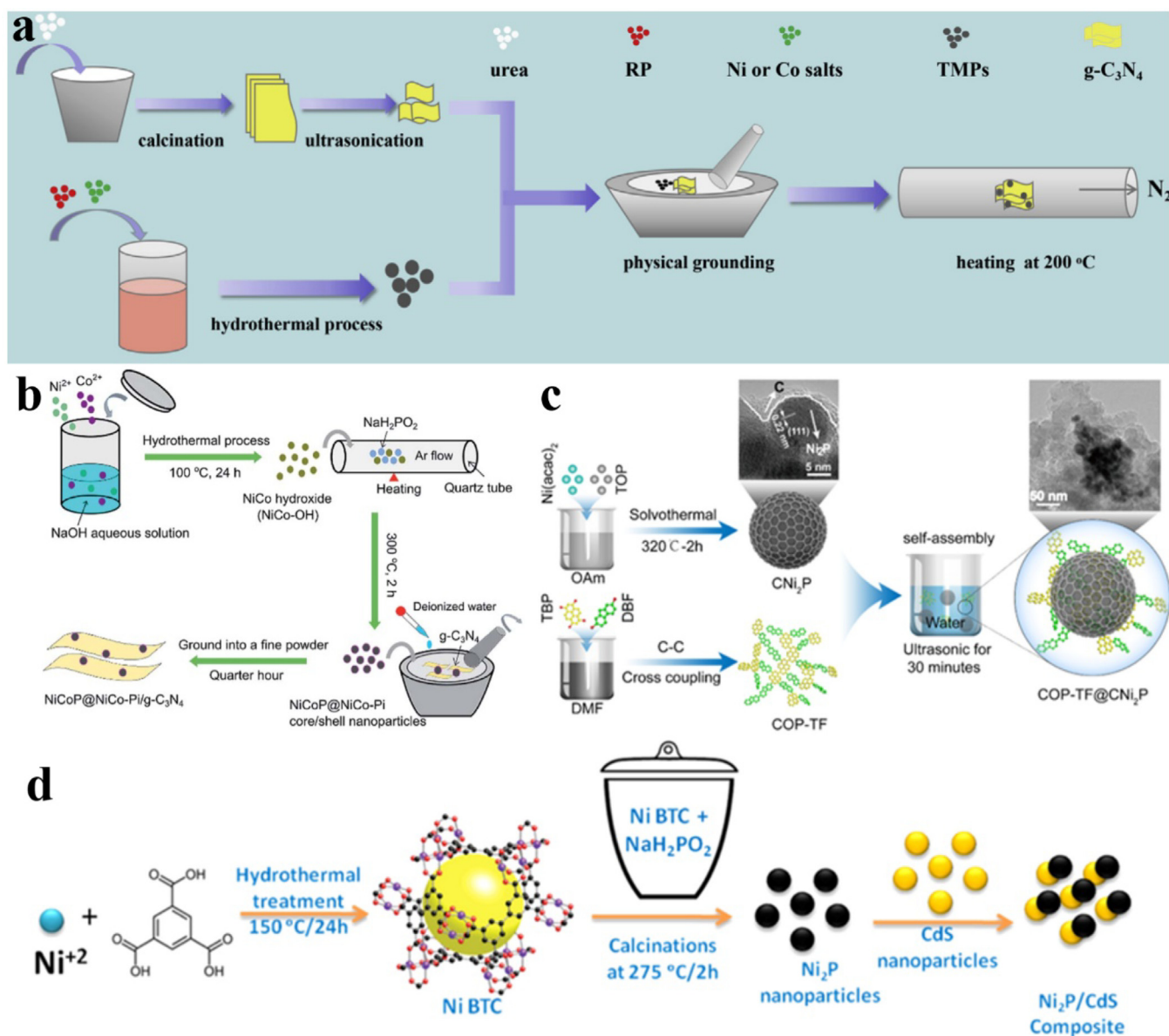
The other strategy is solution-phase synthetic method. Tri-*n*-octylphosphine (TOP) is often utilized as a phosphorous source to prepare TMPs in organic solvent as its C-P covalent bond can be broken at a higher temperature. As a result, metal precursors including metal carbonyl compounds and metal acetylacetonates can be phosphorized in the presence of TOP. For example, as shown in Fig. 4c, Zeng et al. [82] prepared  $\text{Ni}_2\text{P}$  nanoparticles by decomposing  $\text{Ni}(\text{acac})_2$  in oleylamine (OAm) at 210 °C, and subsequently reacting with TOP at 325 °C. However, the reaction is highly flammable and corrosive due to the use of organic solvent and high decomposition temperature of TOP, which hinders its application. Another solution-phase synthetic method is to use the sol-gel chemistry process to produce TMPs from a chemically homogeneous precursor. Generally, it can be completed at shorter reaction times and lower reaction temperature. Furthermore, it can better control the morphology and size of the TMPs. By using  $\text{SiO}_2$  xerogel as the host matrix, Lukehart et al. [83] reported the synthesis of a series of TMPs by a sol-gel method followed by an annealing treatment. MoP/ $\text{SiO}_2$  has also been synthesized using the same method [84].

Recently, hydrothermal/solvothermal method has been employed to synthesize TMPs because of the high efficiency and ease of the experimental process. In this process, a homogeneous solution including phosphorous source, metal source, and solvent is transferred to the Teflon-lined stainless steel autoclave and then reacted at a specified temperature. The phosphorous source was mainly white phosphorous, yellow phosphorous, red phosphorous, and black phosphorous. For example, Fig. 4d demonstrated that red phosphorous (RP) and  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$  could be used as the precursors in the hydrothermal process to synthesize  $\text{Ni}_2\text{P}$  nanoparticles [85]. Sun et al. [86] prepared  $\text{Ni}_2\text{P}$  by a solvothermal reaction at 140 °C using yellow phosphorus as the phosphorous source,  $\text{Ni}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  as the metal source and ethylenediamine as the solvent. Additionally, sonochemical is also an effective solution-phase synthetic method for preparing TMPs, which helps to accelerate the reaction process and tune the morphology of TMPs. For example, FeP could be synthesized from the sonication of a mixture of  $\text{Fe}(\text{CO})_5$  and triethylphosphine solution with a power of 950 W for 300 min followed by the calcination at 950 °C [87]. Microwave-assisted synthesis can obtain TMPs in a shorter time with lower power consumption due to the direct and uniform heating. For example, the preparation of  $\text{Ni}_2\text{P}$  and  $\text{Ni}_{12}\text{P}_5$  only required 2 min of microwave irradiation using tetrabutylphosphonium chloride as phosphorus source and reaction medium [88]. In general, the solution-phase synthetic method is beneficial to control the morphology and size of the TMPs, and the as-prepared TMPs exhibit a higher purity and uniformity. However, the multistep synthesis process and the use of organic solvents may limit its wider applications.

## 4. Loading methods of TMPs on semiconductors

### 4.1. Post loading method

The post loading method is one of the strategies to load the TMPs cocatalysts on semiconductors. In this process, the TMPs are first synthesized and then combined with semiconductors through an extra loading method, such as physical grinding and impregnation. For example, Jin et al. [89] prepared TMPs/g- $\text{C}_3\text{N}_4$  nanocomposites by the



**Fig. 5.** (a) Schematic illustration of the formation of TMPs/g-C<sub>3</sub>N<sub>4</sub> photocatalysts. Reproduced with permission from Ref. [89] Copyright 2019 Elsevier. (b) Schematic illustration for the fabrication of the NiCoP@NiCo-Pi/g-C<sub>3</sub>N<sub>4</sub> photocatalyst. Reproduced with permission from Ref. [90] Copyright 2017 Royal Society of Chemistry. (c) Schematic illustration of the synthetic route of the hybrid photocatalyst COP-TF@CNi<sub>2</sub>P. Reproduced with permission from Ref. [91] Copyright 2019 American Chemical Society. (d) Synthesis of Ni<sub>2</sub>P/CdS Composite. Reproduced with permission from Ref. [92] Copyright 2016 American Chemical Society.

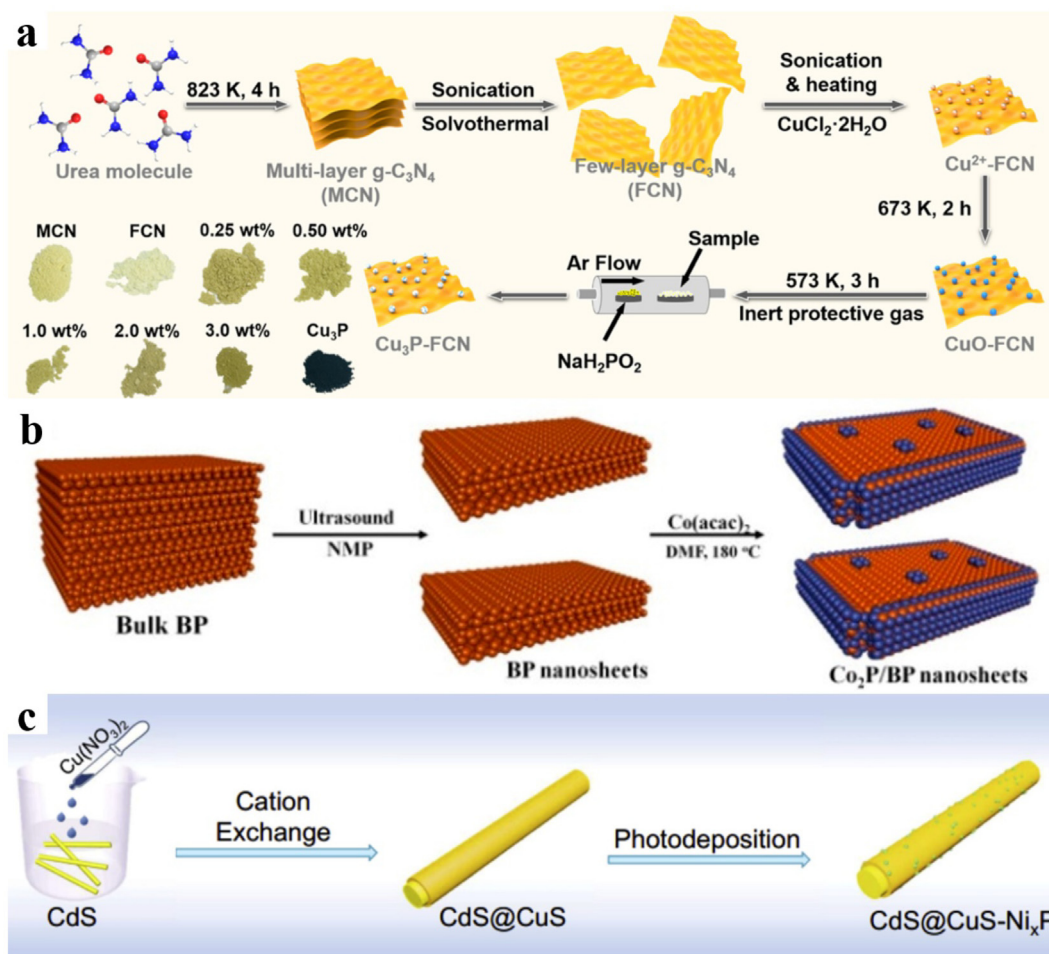
physical grinding followed with annealing. As exhibited in Fig. 5a, the g-C<sub>3</sub>N<sub>4</sub> nanosheets and TMPs were first synthesized by calcination and ultrasonication as well as hydrothermal process, respectively. Then, the g-C<sub>3</sub>N<sub>4</sub> nanosheets and TMPs were mixed and ground in an agate mortar. Finally, the mixture was annealed in N<sub>2</sub> atmosphere at 200 °C to obtain the TMPs/g-C<sub>3</sub>N<sub>4</sub> nanocomposites. Qin et al. [90] also loaded NiCoP@NiCo-Pi cocatalyst on g-C<sub>3</sub>N<sub>4</sub> by the physical grinding. To increase the dispersibility of NiCoP@NiCo-Pi nanoparticles on the g-C<sub>3</sub>N<sub>4</sub> surface, they added deionized water in the mixture during the grinding process, as shown in Fig. 5b.

Furthermore, impregnation method is an effective way to enhance the dispersibility of TMPs on semiconductors. For example, Liu et al. [91] combined carbon-encapsulated nickel phosphide (CNi<sub>2</sub>P) with covalent organic polymer (COP-TF) by an ultrasonic-assisted self-assembly method. As depicted in Fig. 5c, the CNi<sub>2</sub>P and COP-TF were first prepared by solvothermal reaction and copolymerization, respectively. Then the COP-TF@CNi<sub>2</sub>P photocatalyst was obtained by the ultrasonic-

assisted self-assembly method. Because of the separate preparation of the TMPs cocatalysts and semiconductors, TMPs with different morphology and structure can be loaded. Kumar et al. [92] prepared Ni<sub>2</sub>P nanoparticles from porous Ni-MOFs and then deposited it on CdS via an impregnation method, as shown in Fig. 5d. The post loading method is conducive to the control of the morphology and size of TMPs. Nevertheless, the interfacial contact between TMPs cocatalysts and semiconductors is weak due to the separation treatment of this post loading method.

#### 4.2. In situ reduction method

The loading of TMPs cocatalysts on semiconductors can also be performed by the in situ reduction method. This in situ growth method is beneficial for the formation of intimate contact and stable junction between TMPs cocatalysts and semiconductors, thus significantly promoting the transfer of photogenerated charge carriers. Generally, the in



**Fig. 6.** (a) Synthetic procedures for the Cu<sub>3</sub>P-FCN hybrids and digital photographs of the photocatalysts. Reproduced with permission from Ref. [93] Copyright 2019 American Chemical Society. (b) Schematic illustration for the preparation of Co<sub>2</sub>P/BP nanosheets photocatalyst. Reproduced with permission from Ref. [77] Copyright 2019 American Chemical Society. (c) Schematic illustration of the synthesis of the CdS@CuS-Ni<sub>x</sub>P core-shell nanowires. Reproduced with permission from Ref. [96] Copyright 2019 Elsevier.

situ reduction process can be conducted using the chemical vapor deposition (CVD) method, the hydrothermal/solvothermal method and the photochemical reduction method. The CVD method is applicable to the phosphorization of metal, metal oxides and metal hydroxides. For example, Wang et al. [93] fabricated the Cu<sub>3</sub>P/g-C<sub>3</sub>N<sub>4</sub> photocatalyst via a facile CVD method. As shown in Fig. 6a, the few-layer g-C<sub>3</sub>N<sub>4</sub> was prepared through the calcination and solvothermal process at first. Then the CuO was in situ grown on the few-layer g-C<sub>3</sub>N<sub>4</sub>. Finally, the Cu<sub>3</sub>P/g-C<sub>3</sub>N<sub>4</sub> was obtained by phosphorization treatment of CuO/g-C<sub>3</sub>N<sub>4</sub>.

The hydrothermal/solvothermal method is a facile and effective strategy to load TMPs cocatalysts in situ on semiconductors. Yuan et al. [77] reported that Co<sub>2</sub>P could be selectively in situ grown around the edges of BP nanosheets through a simple solvothermal method. As exhibited in Fig. 6b, the few-layer BP nanosheets was first exfoliated from bulk BP in N-methyl-pyrrolidone (NMP) solution with the assistance of ultrasonication. Subsequently, the few-layer BP nanosheets reacted with Co(acac)<sub>3</sub> in N, N-dimethyl formamide (DMF) solution at 180 °C to in situ grow Co<sub>2</sub>P around the edges of BP nanosheets. Besides, the in situ photodeposition of TMPs cocatalysts on semiconductors has also drawn attention because it can be executed at ambient temperature and does not require extra reductant. In this process, the semiconductors absorb light and generate electron-hole pairs under irradiation. Then, part of (H<sub>2</sub>PO<sub>2</sub>)<sup>-1</sup> in the solution are reduced by the photogenerated electrons and react with metal ions to form TMPs [94,95]. For example,

Li et al. [96] deposited Ni<sub>x</sub>P nanoparticles on the CdS@CuS nanowires via a photochemical reduction method, as displayed in Fig. 6c. In short, the in situ reduction method can make the contact between TMPs cocatalysts and semiconductors closer. But the morphology and size of TMPs is difficult to control during this process, which may reduce the number of active sites.

## 5. Application in photocatalytic H<sub>2</sub> production

Among all the studied TMPs, iron phosphides, cobalt phosphides and nickel phosphides are the most frequently used cocatalysts in photocatalytic H<sub>2</sub> production. Besides, copper phosphides, molybdenum phosphides, tungsten phosphides and bimetallic phosphides are also reported to be active in photocatalytic H<sub>2</sub> production as cocatalysts. In the following section, the performances and mechanisms of these TMPs in photocatalytic H<sub>2</sub> production are detailedly introduced and discussed.

### 5.1. Iron phosphides

Owing to their high abundance, extensive distribution, and low price, iron-based materials have been attractive in recent years [97–100]. Moreover, iron-containing clusters are the catalytically active sites in highly efficient biological HER catalysts such as [FeFe] and [Fe]-only hydrogenases [101]. These advantages render iron



phosphides as efficient and inexpensive cocatalysts. And until now, iron phosphides have been extensively investigated and utilized in photocatalytic  $H_2$  evolution system as cocatalysts [102,103]. Iron phosphides usually exist in the form of FeP and  $Fe_2P$ . For example, Callejas et al. [104] immobilized hollow FeP nanoparticles on  $TiO_2$  for photocatalytic  $H_2$  production under ultraviolet light irradiation. The average  $H_2$  production rate and the apparent quantum yield (AQY) at 365 nm for FeP/ $TiO_2$  were  $1900 \mu\text{mol h}^{-1} \text{g}^{-1}$  and 8.7% in methanol aqueous solution, while  $TiO_2$  alone exhibited an ignorable rate of  $H_2$  production, indicating that FeP was the highly active cocatalyst in photocatalytic  $H_2$  production.

Unfortunately, the practical application of FeP/ $TiO_2$  was limited because it could only respond under ultraviolet light, which merely accounts for about 4% of the solar light. Therefore, it is significant to construct novel photocatalysts for  $H_2$  production under visible light irradiation. Cheng et al. [75] found that FeP could improve the photocatalytic performance of CdS under visible light irradiation for  $H_2$  production in lactic acid aqueous solution. The optimized FeP/CdS nanocomposites possessed a high  $H_2$  production rate of  $202000 \mu\text{mol h}^{-1} \text{g}^{-1}$ , which was approximately 67 and 3 times that of pure CdS and Pt/CdS, respectively. The yield of  $H_2$  was dramatically enhanced because FeP could efficiently promote the separation of photogenerated charge from CdS. DFT calculations further clarified the photocatalytic mechanisms between FeP and CdS. When FeP and CdS were in close contact to form a heterojunction, an inherent electric field was built at their interface. This would shift the energy band edges of CdS downwards and the Fermi energy of FeP upwards to reach the equilibrium. The band bending could accelerate electrons transfer from the CdS layer to the FeP layer at the interfacial space-charge region, and thus restrain the recombination of photogenerated electron-hole pairs. Meanwhile, the formation of the Schottky barrier also significantly promoted the spatial charge separation, resulting in the improved photocatalytic activity for  $H_2$  production.

Although FeP could improve the photocatalytic  $H_2$  production performance of CdS, the morphology of FeP/CdS nanocomposites in previous studies was mainly nanoparticles. The nanoparticles are easy to self-aggregate, which will result in a reduction in the number of active sites in photocatalytic reactions. Immobilizing cocatalysts nanoparticles on semiconductors nanosheets can efficiently resolve this issue, which not only increase the dispersion of cocatalysts but also promote the separation and transfer of photogenerated charge over semiconductors nanosheets. Sun et al. [105] fixed zero-dimensional (0D) FeP nanoparticles into two-dimensional (2D) CdS nanosheets for efficient photocatalytic  $H_2$  production. The 0D FeP nanoparticles were uniformly dispersed on the surface of 2D CdS nanosheets and a tight interfacial interaction was established on them. This intimate interfacial contact would facilitate the photogenerated charge transfer and separation, consequently increasing the yield of  $H_2$ . Specifically, under visible light irradiation, electrons in the valence band of 2D CdS nanosheets were excited to the CB, leaving holes in the valence band. Subsequently, photogenerated electrons accumulated on the CB migrated to the surface of 0D FeP nanoparticles through the close interfacial contact and reduced protons to  $H_2$ . Moreover, 2D  $g\text{-C}_3\text{N}_4$  nanosheets was considered a good platform for dispersing 0D nanomaterials [106]. Zeng et al. [107] achieved superior photocatalytic  $H_2$  production by uniformly depositing ultrasmall FeP nanodots on the 2D porous  $g\text{-C}_3\text{N}_4$  nanosheets. Benefiting from the close contact between FeP and  $g\text{-C}_3\text{N}_4$ , the 0D/2D FeP/ $g\text{-C}_3\text{N}_4$  presented accelerated photogenerated charge separation ability and enhanced photocatalytic  $H_2$  production activity. The optimized FeP/ $g\text{-C}_3\text{N}_4$  sample had a maximum  $H_2$  production rate of  $177.9 \mu\text{mol h}^{-1} \text{g}^{-1}$  with the AQY of 1.57% at 420 nm.

In addition to FeP,  $Fe_2P$  was also employed as a cocatalyst for photocatalytic  $H_2$  production owing to its good electronic conductivity [108,109]. Sun et al. [110] reported that the photocatalytic  $H_2$  production rate could be enhanced by > 30 times by anchoring of  $Fe_2P$

nanoparticles on the surface of CdS nanorods. The enhanced activity was ascribed to the efficient transfer of photogenerated electrons from the CdS to the  $Fe_2P$  through the metal-semiconductor interfaces, facilitating the photogenerated charge separation. Meanwhile, the  $Fe_2P$  passivated the defects on the CdS surface, which made the electrons energetically favorable for photocatalytic  $H_2$  production. Furthermore, Zhao et al. [54] prepared FeP and  $Fe_2P$  co-modified  $g\text{-C}_3\text{N}_4$  photocatalyst for  $H_2$  production under visible light irradiation. During the reaction, one proton in  $H_2O$  molecule was firstly adsorbed on the Fe ( $\delta^+$ ) and P ( $\delta^-$ ), respectively. Then the electrons in H-O bond migrated to the O atom to generate a dual protonation transition state and release  $OH^-$ . Finally, as the photogenerated electrons transfer from  $g\text{-C}_3\text{N}_4$  to Fe ( $\delta^+$ ) atom, the hydride was formed around the Fe ( $\delta^+$ ) center, and further combined with the proton around the P ( $\delta^-$ ) to generate a  $H_2$  molecule. As a result, because the  $Fe_xP$  cocatalyst could dramatically promote the separation and transfer of photogenerated electrons on  $g\text{-C}_3\text{N}_4$  and the adjacent Fe and P atoms in  $Fe_xP$  could serve as dual proton adsorption sites, the  $H_2$  production rate of optimized  $Fe_xP/g\text{-C}_3\text{N}_4$  was 277 times higher than that of pristine  $g\text{-C}_3\text{N}_4$ .

Based on the above research results, it is obviously that the iron phosphides cocatalysts can efficiently improve the activity of photocatalytic  $H_2$  production. But in the current study, the morphology of most iron phosphides is nanoparticles, which are easy to self-aggregate, leading to a decrease in the number of active sites in photocatalytic  $H_2$  production. Accordingly, further studies should focus on the development of various nanostructured iron phosphides with abundant active sites, such as 2D ultrathin nanosheets and 0D quantum dots.

## 5.2. Cobalt phosphides

Various cobalt phosphides as excellent cocatalysts for photocatalytic  $H_2$  production have stimulated great interest (Table 2). Among these, CoP was investigated mostly, and a series of CoP-based photocatalysts were fabricated for photocatalytic  $H_2$  production, such as CoP/ $Cd_xZn_{1-x}Se$  [111], CoP/BP [112], CoP/ $Zn_{0.5}Cd_{0.5}S$  [113], CoP/MIL-125- $NH_2$  [114], CoP/ $CeVO_4$  [115], CdS@CoP@ $SiO_2$  [116] and CdS/RGO-MoS<sub>2</sub>@CoP [117]. For example, Yue et al. [118] found that CoP could significantly accelerate the photogenerated charge transfer and inhibit the photogenerated electron-hole pairs recombination of  $TiO_2$ , thereby enhancing the photocatalytic  $H_2$  production rate. The  $H_2$  production rate of the optimized CoP/ $TiO_2$  was  $8350 \mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 11 times higher than that of the pristine  $TiO_2$ . Yi et al. [119] reported that the introduction of CoP led to a remarkable enhancement in the photocatalytic  $H_2$  production activity of  $g\text{-C}_3\text{N}_4$ . Compared to pure  $g\text{-C}_3\text{N}_4$  ( $3.6 \mu\text{mol h}^{-1} \text{g}^{-1}$ ), the resultant CoP/ $g\text{-C}_3\text{N}_4$  presented superior photocatalytic  $H_2$  production rate of  $474.4 \mu\text{mol h}^{-1} \text{g}^{-1}$ . Tan et al. [120] loaded CoP on La, Cr: SrTiO<sub>3</sub> to obtain a novel photocatalyst with improved  $H_2$  production activity. Due to the extended visible light response range and improved photogenerated charge transfer and separation, the photocatalytic  $H_2$  production rate of the optimized CoP/La, Cr: SrTiO<sub>3</sub> was 27 times higher than that of La, Cr: SrTiO<sub>3</sub>, reaching  $198.4 \mu\text{mol h}^{-1} \text{g}^{-1}$ .

In addition to CoP, other phases of cobalt phosphides including Co<sub>2</sub>P, CoP<sub>3</sub> and amorphous Co<sub>x</sub>P have also drawn much attention. Li et al. [128] evenly dispersed Co<sub>2</sub>P nanoparticles on the surface of CdS sub-microspheres via an in situ hydrothermal method. The optimized Co<sub>2</sub>P/CdS possessed a photocatalytic  $H_2$  production rate of  $6060 \mu\text{mol h}^{-1} \text{g}^{-1}$ , which was about 35 times higher than that of pure CdS. The enhanced photocatalytic activity resulted from the accelerated separation of photogenerated charge promoted by the proper band bending between Co<sub>2</sub>P and CdS, as verified by the surface photovoltage (SPV) spectra and DFT calculation. Huang et al. [55] constructed a novel CoP<sub>3</sub>/Mn<sub>0.2</sub>Cd<sub>0.8</sub>S nanocomposite, in which CoP<sub>3</sub> nanoparticles were homogeneously and tightly dispersed on the surface of Mn<sub>0.2</sub>Cd<sub>0.8</sub>S nanowires. The time-resolved photoluminescence (TRPL) spectra demonstrated that the introduction of CoP<sub>3</sub> prolonged the



**Table 2**  
Cobalt phosphides cocatalysts for photocatalytic H<sub>2</sub> production.

Photocatalyst	Cocatalyst	Light source	Sacrificial agents	Activity( $\mu\text{mol h}^{-1} \text{g}^{-1}$ )	AQY (%) $\lambda$ (nm)	Ref. (year)
CoP/g-C <sub>3</sub> N <sub>4</sub>	CoP	UV-Vis (Xe)	Methanol	1038.1	1.10 (380)	[121] (2019)
CoP/TiO <sub>2</sub>	CoP	UV-Vis (Xe)	TEOA	8350	3.80 (350)	[118] (2017)
CoP/Zn <sub>0.5</sub> Cd <sub>0.5</sub> S	CoP	AM 1.5G (Xe)	Ascorbic acid	12175.8	4.37 (420)	[122] (2018)
CoP/CdS/WS <sub>2</sub>	CoP	$\lambda > 420 \text{ nm}$ (Xe)	—	9.16	1.34 (420)	[78] (2019)
CoP/P-g-C <sub>3</sub> N <sub>4</sub>	CoP	$\lambda > 420 \text{ nm}$ (Xe)	Methanol	724	8.5 (420)	[123] (2020)
CoP/g-C <sub>3</sub> N <sub>4</sub>	CoP	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	956.8	3.65 (420)	[58] (2019)
CoP/g-C <sub>3</sub> N <sub>4</sub>	CoP	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	1074	6.10 (420)	[124] (2019)
CoP/BCNDs/CNNS	CoP	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	1332.81	10.78 (420)	[125] (2019)
CoP/CdS	CoP	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	13,785	11.60 (420)	[126] (2018)
CdS@CoP@SiO <sub>2</sub>	CoP	$\lambda > 420 \text{ nm}$ (Xe)	LA	17,727	23.60 (435)	[116] (2018)
Co <sub>2</sub> P/BP	Co <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	202	2.42 (420)	[77] (2019)
Co <sub>2</sub> P/CdS	Co <sub>2</sub> P	$\lambda > 780 \text{ nm}$ (MH)	Lactic acid	3930	2.26 (700)	[127] (2019)
Co <sub>2</sub> P/CdS	Co <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	LA + K <sub>2</sub> HPO <sub>4</sub>	71,200	13.88 (420)	[128] (2018)
o-Co <sub>2</sub> P/CdS	Co <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	LA	184,480	22.17 (420)	[129] (2019)
CoP <sub>3</sub> /Mn <sub>0.2</sub> Cd <sub>0.8</sub> S	CoP <sub>3</sub>	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	29,530	29.20 (420)	[55] (2018)
Co <sub>1</sub> -phosphide/PCN	CoP <sub>4</sub>	$\lambda > 420 \text{ nm}$ (Xe)	—	126.8	2.20 (500)	[130] (2017)

AQY: apparent quantum yield; Xe: xenon lamp; MH: metal halide lamp; TEOA: triethanolamine; LA: lactic acid.

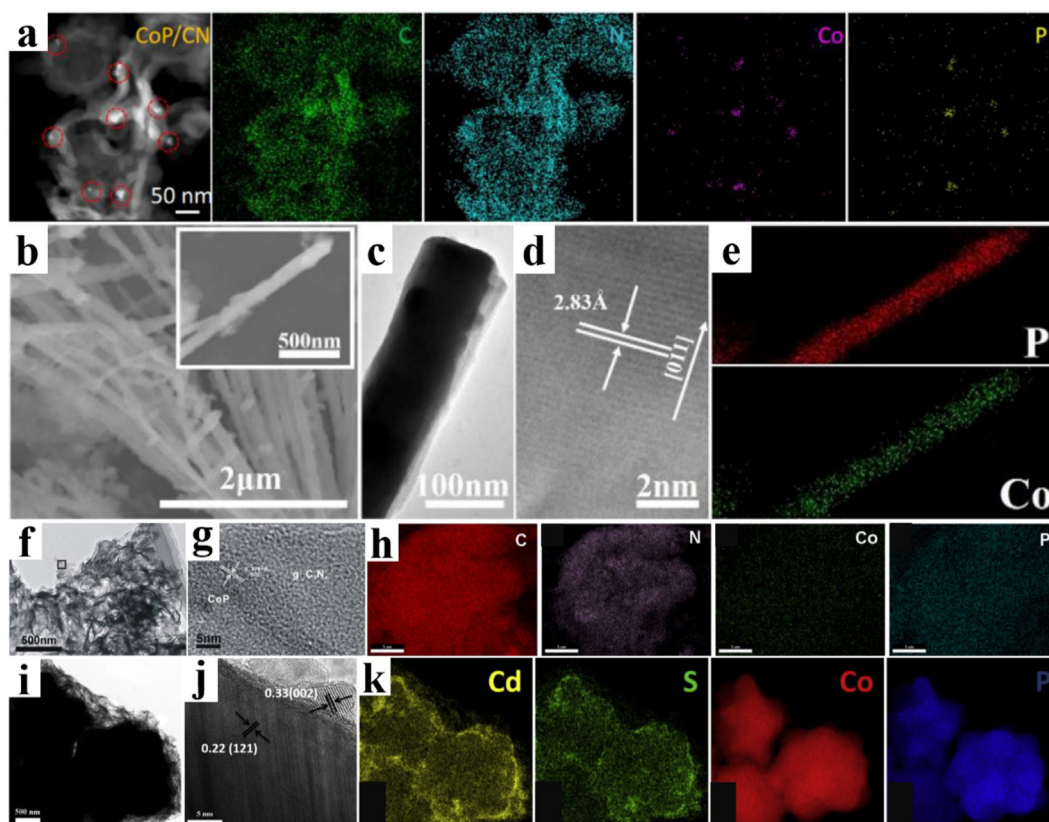
photogenerated charge lifetime of Mn<sub>0.2</sub>Cd<sub>0.8</sub>S (from 0.5256 to 0.8697 ns), thus promoting more photogenerated electrons participating in the photocatalytic H<sub>2</sub> production reaction. Accordingly, the superior H<sub>2</sub> production rate of 29530  $\mu\text{mol h}^{-1} \text{g}^{-1}$  was achieved on CoP<sub>3</sub>/Mn<sub>0.2</sub>Cd<sub>0.8</sub>S under visible light irradiation, which was 5.02 times than that of pure Mn<sub>0.2</sub>Cd<sub>0.8</sub>S. Moreover, the amorphous CoP<sub>x</sub> shell was loaded on CdS nanorods for photocatalytic H<sub>2</sub> production by Sun and coworkers [131]. The optimized CoP<sub>x</sub>/CdS presented a H<sub>2</sub> production rate of 204000  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was much higher than that of pure CdS (25000  $\mu\text{mol h}^{-1} \text{g}^{-1}$ ). The AQY was measured to be 35% after 5 h of irradiation at 450 nm, suggesting the excellent photocatalytic H<sub>2</sub> production performance of CoP<sub>x</sub>/CdS.

Despite enhanced photocatalytic H<sub>2</sub> production activity was achieved by introducing cobalt phosphides cocatalyst, there is still room for improvement. It has been reported that the nanostructure engineering could boost the HER performance of catalysts [132]. Therefore, cobalt phosphides with different morphologies were developed and utilized as cocatalysts in photocatalytic H<sub>2</sub> production, such as 0D quantum dots [133] and nanoparticles [126], one-dimensional (1D) nanowires [134] and nanorods [135], 2D nanosheets [136] and three-dimensional (3D) microspheres [137]. Luo et al. [124] incorporated 0D CoP nanoparticles into 2D g-C<sub>3</sub>N<sub>4</sub> nanosheets through the electrostatic driven self-assembly method and phosphorization method. Because the well dispersed 0D CoP nanoparticles (Fig. 7a) could provide more reaction active sites than the bulk CoP, the 0D/2D CoP/g-C<sub>3</sub>N<sub>4</sub> nanohybrids exhibited a superior photocatalytic H<sub>2</sub> production rate of 1074  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 41 times higher than that of the same mass bulk CoP modified g-C<sub>3</sub>N<sub>4</sub>. However, the 0D nanoparticles are difficult to be uniformly loaded on the surface of photocatalysts, and they tend to agglomerate and form large clusters.

One-dimensional nanostructured materials are believed to hold great application potential in the field of solar cells, electronic devices and photocatalysis due to their high aspect ratio and excellent electron transport property [138,139]. Recently, 1D CoP nanowires with diameters about 100 nm were prepared via a low-temperature phosphidation of Co<sub>3</sub>O<sub>4</sub> nanowires by Wang et al. [122], as displayed in Fig. 7b-d. The elemental mapping images in Fig. 7e further demonstrated that the elements of P and Co were uniformly distributed in the whole CoP nanowire. By introducing 1D CoP nanowires to Zn<sub>0.5</sub>Cd<sub>0.5</sub>S, the outstanding photocatalytic H<sub>2</sub> production activity was achieved (12175.8  $\mu\text{mol h}^{-1} \text{g}^{-1}$ ), which was 22 and 2 times higher than that of the pure Zn<sub>0.5</sub>Cd<sub>0.5</sub>S and the CoP nanoparticles modified Zn<sub>0.5</sub>Cd<sub>0.5</sub>S, respectively. The enhanced activity was attributed to the synergistic effect of the accelerated charge separation and transfer from Zn<sub>0.5</sub>Cd<sub>0.5</sub>S to CoP nanowires and the rapid H<sub>2</sub> production on 1D nanostructure.

Meanwhile, 2D nanomaterials have also received great attention for their exotic electronic properties and high specific surface areas. Wang et al. [62] designed a 2D/2D nanostructure constructed by CoP nanosheets and g-C<sub>3</sub>N<sub>4</sub> nanosheets to expand the Schottky effect between their interfaces. As displayed in Fig. 7f-g, the CoP nanosheets were effectively combined with the g-C<sub>3</sub>N<sub>4</sub> nanosheets, forming a well dispersed sheet-on-sheet structure. Both Co and P elements were homogeneously distributed throughout the 2D/2D CoP/g-C<sub>3</sub>N<sub>4</sub> nanocomposite (Fig. 7h), suggesting the inter-growth of the CoP nanosheets on the g-C<sub>3</sub>N<sub>4</sub> nanosheets. Benefiting from the enhanced interaction, reduced self-agglomeration, enlarged Schottky effect, increased active sites, and shortened carrier transfer distances, the AQY of the 2D/2D CoP/g-C<sub>3</sub>N<sub>4</sub> heterojunction was 2.1 times that of the corresponding 0D/2D heterojunction. In comparison with 0D, 1D and 2D nanostructures, 3D assemblies are more attractive owing to their unique architecture and properties. He et al. [137] demonstrated that 3D Co<sub>2</sub>P microspheres were a good platform to load CdS nanowires (Fig. 7i-k), which uniformly dispersed on the surface of 3D Co<sub>2</sub>P microspheres presented facilitated photogenerated charge separation and enhanced photocatalytic H<sub>2</sub> production activity.

Although a relatively high photogenerated charge carrier separation efficiency and high photocatalytic H<sub>2</sub> production activity were obtained in these photocatalysts, the understanding of the interfacial interactions and the formed chemical bonds between cobalt phosphides cocatalysts and semiconductor photocatalysts is still insufficient. In the past few years, only a few reports have explored the relationship between chemical bonds and photocatalytic performance. For example, Yuan et al. [77] found that the strong Co-P chemical bonds were formed between Co<sub>2</sub>P and BP on the Co<sub>2</sub>P/BP nanosheets photocatalyst. DFT calculations (Fig. 8a) demonstrated that the Co-P bonds could efficiently improve the photogenerated charge carrier transfer between the Co<sub>2</sub>P layer and the BP layer in the atomic level. Therefore, the Co<sub>2</sub>P/BP nanosheets photocatalyst presented an enhanced photocatalytic H<sub>2</sub> generation rate, which was 39.7 times higher than that of pure BP nanosheets. Moreover, the Co-S bonds in Co<sub>2</sub>P/CdS photocatalyst were regarded as a Z-scheme “bridge” for charge transfer (Fig. 8b), which could significantly promote the separation of photogenerated electron-hole pairs and improve the photocatalytic H<sub>2</sub> production activity [127]. To deeply understand the chemical bonds and intrinsic active sites at atomic level, atomically dispersed catalysts have been developed in recent years. Liu et al. [130] confined a single Co<sub>1</sub>-P<sub>4</sub> site on g-C<sub>3</sub>N<sub>4</sub> nanosheets via a facile phosphidation strategy (Fig. 8c). Correlated atomic characterizations corroborated that atomically dispersed Co atoms were successfully anchored by covalently forming an isolated Co<sub>1</sub>-P<sub>4</sub> structure on g-C<sub>3</sub>N<sub>4</sub> nanosheets. The atomically dispersed Co<sub>1</sub>-phosphide active site could significantly inhibit charge recombination



**Fig. 7.** (a) The TEM images and elemental mapping results of CoP/CN with element C, N, Co, and P. Reproduced with permission from Ref. [124] Copyright 2019 Elsevier. (b) SEM, (c) TEM, (d) HRTEM and (e) elemental mapping images of CoP nanowires. Reproduced with permission from Ref. [122] Copyright 2018 Elsevier. (f) TEM, (g) HRTEM and (h) SEM mapping images of 2%CoP/CN composite. Reproduced with permission from Ref. [62] Copyright 2018 Royal Society of Chemistry. (i) TEM, (j) HRTEM and (k) elemental mapping images of CdS-15/Co<sub>2</sub>P microspheres. Reproduced with permission from Ref. [137] Copyright 2019 Elsevier.

and prolong carrier lifetime by about 20 times relative to pristine g-C<sub>3</sub>N<sub>4</sub>, and promote H<sub>2</sub> production. Accordingly, the Co<sub>3</sub>-phosphide/PCN photocatalyst owned exceptionally high photocatalytic activity, with H<sub>2</sub> production rates as high as 410.3  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 23 and 7.6 times higher than those of PCN and CoP cluster/PCN, respectively.

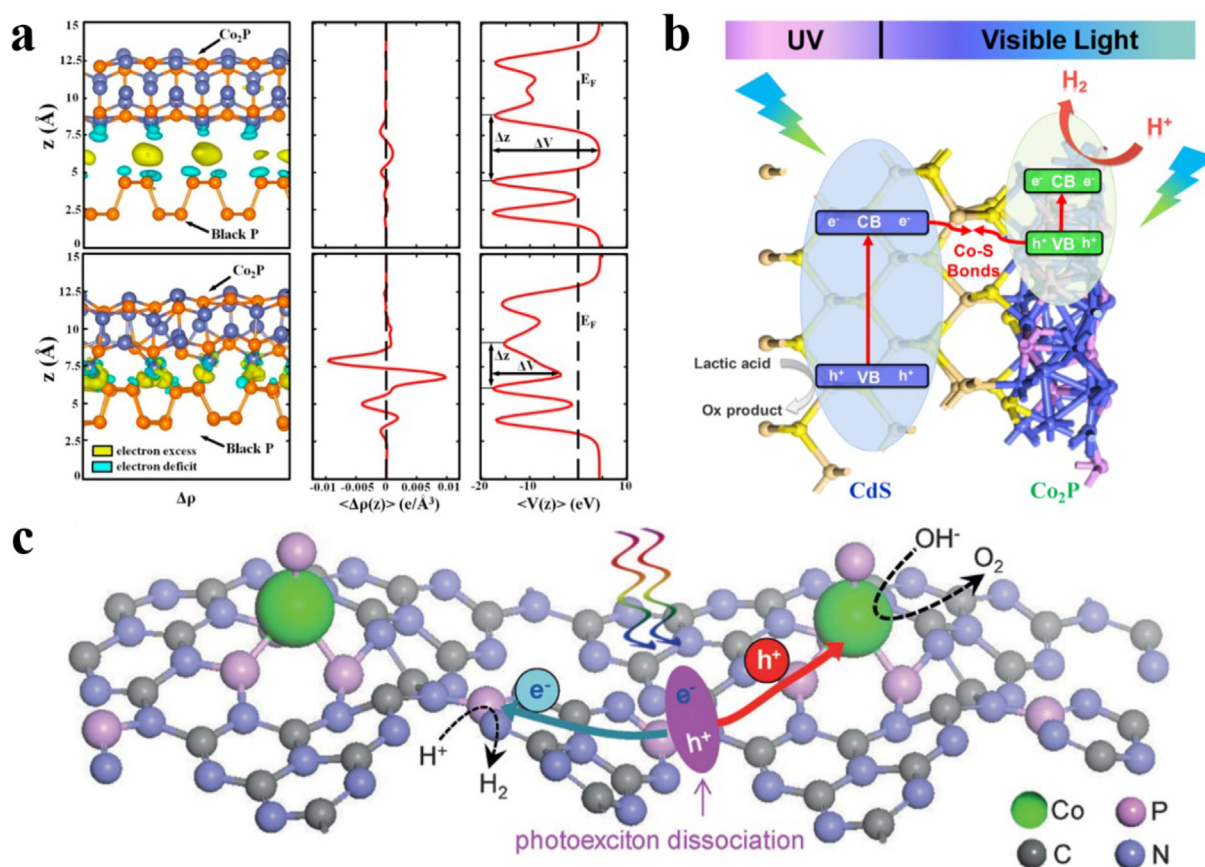
Apart from photocatalytic water splitting, photocatalytic formic acid (FA) dehydrogenation has also been considered as a promising method for H<sub>2</sub> production. Ultrasmall CoP nanoparticle was introduced as an efficient and robust cocatalyst for photocatalytic FA dehydrogenation by Cao and coworkers [140]. As shown in the DFT simulation in Fig. 9a-d, in comparison with noble-metal Pt, CoP presented lower H<sub>2</sub> desorption energy in the H<sub>2</sub> production process, which would be beneficial for the H<sub>2</sub> production. When compounded with CdS@RGO, the hybrid possessed an excellent photocatalytic FA dehydrogenation rate of  $182000 \pm 12500 \mu\text{mol h}^{-1} \text{g}^{-1}$ , which was > 30 and 3 times higher than that of pure CdS and Pd/g-C<sub>3</sub>N<sub>4</sub> photocatalysts, respectively. Moreover, Zhou et al. [141] implanted an atomically dispersed Co-P<sub>3</sub> species on CdS nanorods (CoPSA-CdS) for FA dehydrogenation under visible light irradiation. As revealed by the in situ attenuated total reflection infrared (ATR-IR) spectra (Fig. 9e-g), the CoPSA-CdS presented much better ability for FA dissociation adsorption and C-H bond activation than phosphorus-modified CdS (P-CdS) and sulfur-coordinated Co single atom-loaded CdS nanorods (CoSSA-CdS). DFT calculations (Fig. 9h-i) further verified that the Co-P<sub>3</sub> species accelerated the dissociation adsorption of FA by forming an active P-HCOO intermediate. Therefore, CoPSA-CdS photocatalyst exhibited an excellent photocatalytic activity for the dehydrogenation of FA to H<sub>2</sub>.

The aforementioned investigations demonstrate that cobalt phosphides is promising cocatalysts for photocatalytic H<sub>2</sub> production, which

is ascribed to their easily adjustable nanostructures. Nonetheless, the nanostructured cobalt phosphides are usually loaded on semiconductors by the post loading method, weakening the interfacial contact between cobalt phosphides and semiconductors. Therefore, future investigations should focus on how to make the nanostructured cobalt phosphides contact tightly with the semiconductors to decrease the Schottky barrier.

### 5.3. Nickel phosphides

In the family of nickel phosphides, Ni<sub>2</sub>P and Ni<sub>12</sub>P<sub>5</sub> were the most studied materials as cocatalysts for photocatalytic H<sub>2</sub> production due to the excellent activity and stability (Table 3). For example, Wang et al. [142] anchored Ni<sub>2</sub>P nanocrystals onto the g-C<sub>3</sub>N<sub>4</sub> nanosheets through P-N chemical bonding. The presence of Ni<sub>2</sub>P trapped the photo-generated electrons via a Z-scheme mechanism, thereby remarkably facilitating the separation of photogenerated electron-hole pairs and subsequent reduction of protons to produce H<sub>2</sub>. The optimized Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> nanocomposite showed improved photocatalytic H<sub>2</sub> production rate of 362.4  $\mu\text{mol h}^{-1} \text{g}^{-1}$  under visible light irradiation, which was about 22 times higher than that of pure g-C<sub>3</sub>N<sub>4</sub>. Zeng et al. [143] embedded Ni<sub>12</sub>P<sub>5</sub> nanoparticles into porous g-C<sub>3</sub>N<sub>4</sub> nanosheets to enhance the photocatalytic H<sub>2</sub> production performance. The Ni<sub>12</sub>P<sub>5</sub> nanoparticles could promote the transfer of photogenerated charges and serve as an active site for H<sub>2</sub> production. Benefiting from the accelerated charge transfer and increased catalytic site, the Ni<sub>12</sub>P<sub>5</sub>/g-C<sub>3</sub>N<sub>4</sub> photocatalyst exhibited a superior photocatalytic activity for H<sub>2</sub> production. Moreover, Zhao et al. [144] reported that Ni<sub>3</sub>P could be utilized as a cocatalyst for efficient photocatalytic H<sub>2</sub> production because of its excellent trapping-electron ability. To explore the phase effect of



**Fig. 8.** (a) Electronic structure changes associated with assembling the Co<sub>2</sub>P/BP heterojunction without (upper panel) and with (middle panel) chemical bonding from isolated layers. Reproduced with permission from Ref. [77] Copyright 2019 American Chemical Society. (b) Illustration of mechanism of Co<sub>2</sub>P/CdS under UV–vis light irradiation. Reproduced with permission from Ref. [127] Copyright 2019 American Chemical Society. (c) Schematic illustration of the solar-driven overall water splitting on the Co<sub>1</sub>-phosphide/PCN photocatalyst. Reproduced with permission from Ref. [130] Copyright 2017 Wiley.

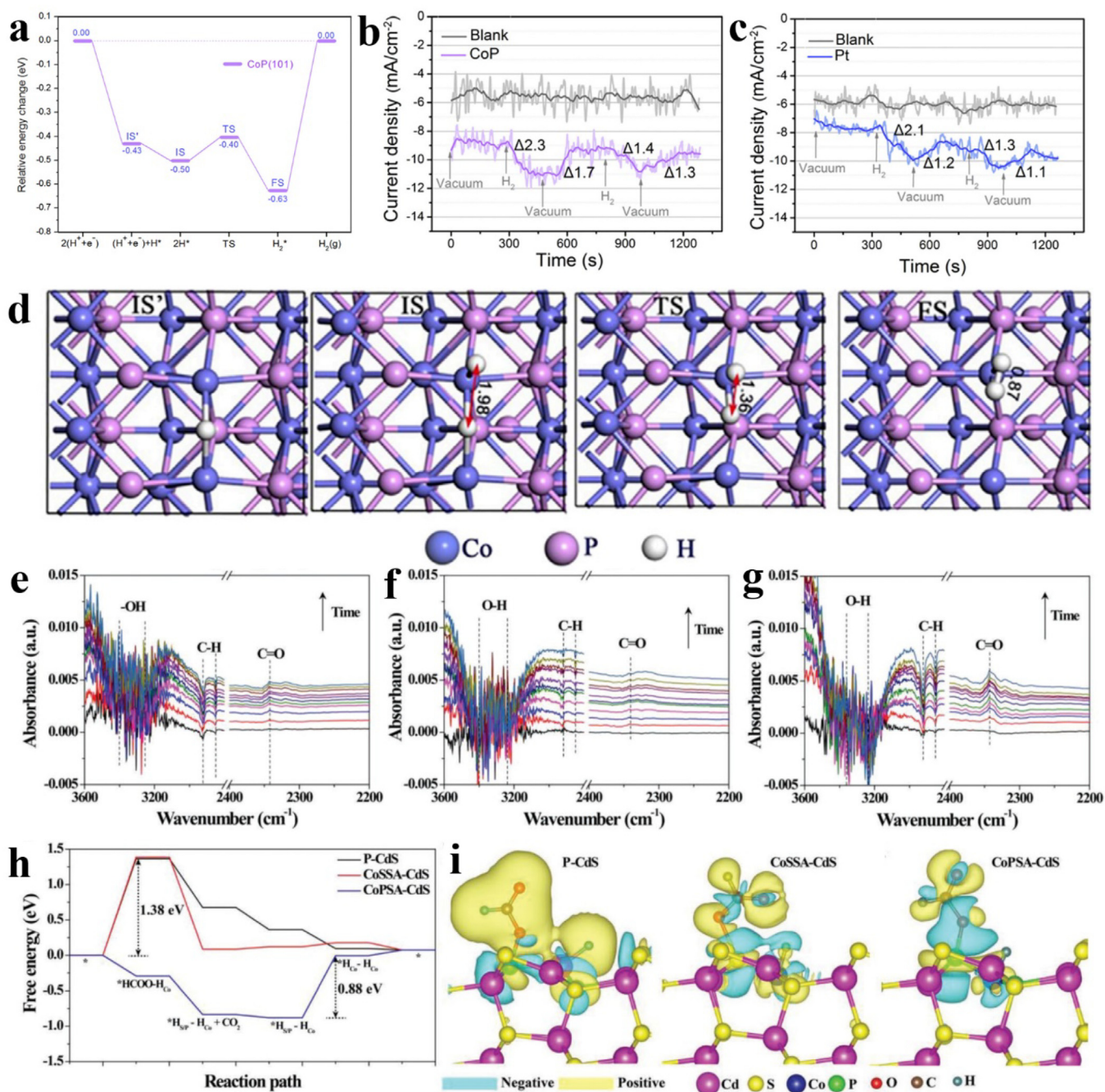
nickel phosphides on photocatalytic H<sub>2</sub> production, three different phases of nickel phosphides (Ni<sub>2</sub>P, Ni<sub>12</sub>P<sub>5</sub>, and Ni<sub>3</sub>P) were prepared and then combined with g-C<sub>3</sub>N<sub>4</sub> [56]. All three phases of nickel phosphides could efficiently enhance the photocatalytic activity of g-C<sub>3</sub>N<sub>4</sub> for H<sub>2</sub> production, with the most significant effect of Ni<sub>2</sub>P. This was because Ni<sub>2</sub>P had a higher ratio of phosphorus, which could promote charge transfer and provide more Ni–P bonds, resulting in a superior H<sub>2</sub> production ability.

In order to further enhance the transfer ability of photogenerated charge carrier, a series of unique nanostructures have been designed to strengthen the interfacial contact between Ni<sub>2</sub>P cocatalyst and semiconductor photocatalysts [156–158]. Zeng et al. [82] anchored monodisperse sub-15 nm Ni<sub>2</sub>P nanoparticles on porous g-C<sub>3</sub>N<sub>4</sub> nanosheets to fabricate 0D/2D heterojunction interfaces (Fig. 10a). The 2D structure shortened the transfer distance of photogenerated charge carriers, and the Ni<sub>2</sub>P accelerated the charge separation and transfer to inhibit the recombination of electron–hole pairs. Benefiting from the unique properties of 2D g-C<sub>3</sub>N<sub>4</sub> nanosheets and 0D Ni<sub>2</sub>P nanoparticles as well as the intimate contact between them, the 0D/2D Ni<sub>2</sub>P/g-C<sub>3</sub>N<sub>4</sub> nanocomposite presented significant enhanced photocatalytic H<sub>2</sub> production activity under visible light irradiation. Zhen et al. [68] constructed a core–shell structured Ni<sub>2</sub>P@CdS photocatalyst for achieving highly efficient interfacial contact. As shown in Fig. 10b, the Ni<sub>2</sub>P@CdS was a core–shell structure with a size of about 50 nm. Under visible light irradiation, the photogenerated electrons on the CB of CdS were migrated to the shell of Ni<sub>2</sub>P. The shell of Ni<sub>2</sub>P acted as both the platform to catch electrons and the reduction active site of proton to H<sub>2</sub>. Therefore, the optimized Ni<sub>2</sub>P@CdS photocatalyst exhibited a superior photocatalytic H<sub>2</sub> production rate of 838  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , revealing about 28.7-fold

enhancement compared to that of the 1Pt@CdS. Furthermore, Boppella et al. [146] combined a 2D Ni<sub>2</sub>P@BP cocatalyst with a 2D porous g-C<sub>3</sub>N<sub>4</sub> nanosheets to improve the photocatalytic H<sub>2</sub> production. Fig. 10c demonstrated that a close interfacial contact was obtained between Ni<sub>2</sub>P@BP nanosheets and g-C<sub>3</sub>N<sub>4</sub> nanosheets. The BP nanosheets prevented the agglomeration of Ni<sub>2</sub>P and build a strong ingenious interface with g-C<sub>3</sub>N<sub>4</sub> nanosheets. Meanwhile, the uniformly distributed Ni<sub>2</sub>P improved conductivity of the photocatalyst and provided abundant active sites for H<sub>2</sub> production. By virtue of these merits, the optimized 2D Ni<sub>2</sub>P@BP/g-C<sub>3</sub>N<sub>4</sub> photocatalyst displayed an excellent photocatalytic H<sub>2</sub> production activity of 858.2  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was about 50 times higher than that of the pure g-C<sub>3</sub>N<sub>4</sub>.

Besides, Ni<sub>2</sub>P was also utilized in some complex photocatalytic H<sub>2</sub> production systems to meet the demand of practical application. Although 97% of the water resource on the earth is seawater, effective photocatalytic H<sub>2</sub> production from seawater is still challenging due to the existence of complex and diverse cations and microorganisms. Liu et al. [91] loaded carbon-encapsulated Ni<sub>2</sub>P on a fully conjugated organic polymer (COP-TF@CNi<sub>2</sub>P) for stable and efficient photocatalytic H<sub>2</sub> production from seawater splitting. As exhibited in Fig. 10d, the optimized COP-TF@CNi<sub>2</sub>P showed an excellent photocatalytic H<sub>2</sub> production rate of 2500  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 10-fold higher than COP-TF@Pt from seawater. Moreover, the COP-TF@CNi<sub>2</sub>P still kept 92% of photocatalytic efficiency after 16 intermittent cycles in seawater conditions, which lasted for half a month (Fig. 10e). The outstanding activity and long-term stability were owing to the compact integration between COP-TF and CNi<sub>2</sub>P, which accelerated the transfer of photogenerated electrons to the surface of CNi<sub>2</sub>P and resisted shedding during photocatalytic H<sub>2</sub> production. Recently, it has been found that





**Fig. 9.** (a) Calculated energy profile for hydrogen production on CoP (101) surface; Gas adsorption–desorption test for (b) CoP and (c) Pt coated electrode at 1 V; (d) The optimized structures of the initial state (IS), transition state (TS), and final state (FS), as labeled in (a). Reproduced with permission from Ref. [140] Copyright 2018 Elsevier. In situ IR spectrum analysis of the photocatalytic FA dehydrogenation over (e) P-CdS, (f) CoSSA-CdS and (g) CoPSA-CdS; (h) The calculated energy profile for FA dehydrogenation and hydrogen production on P-CdS, CoSSA-CdS and CoPSA-CdS; (i) The charge density difference maps between the adsorbed FA and CdS of P-CdS, CoSSA-CdS and CoPSA-CdS. Reproduced with permission from Ref. [141] Copyright 2019 Wiley.

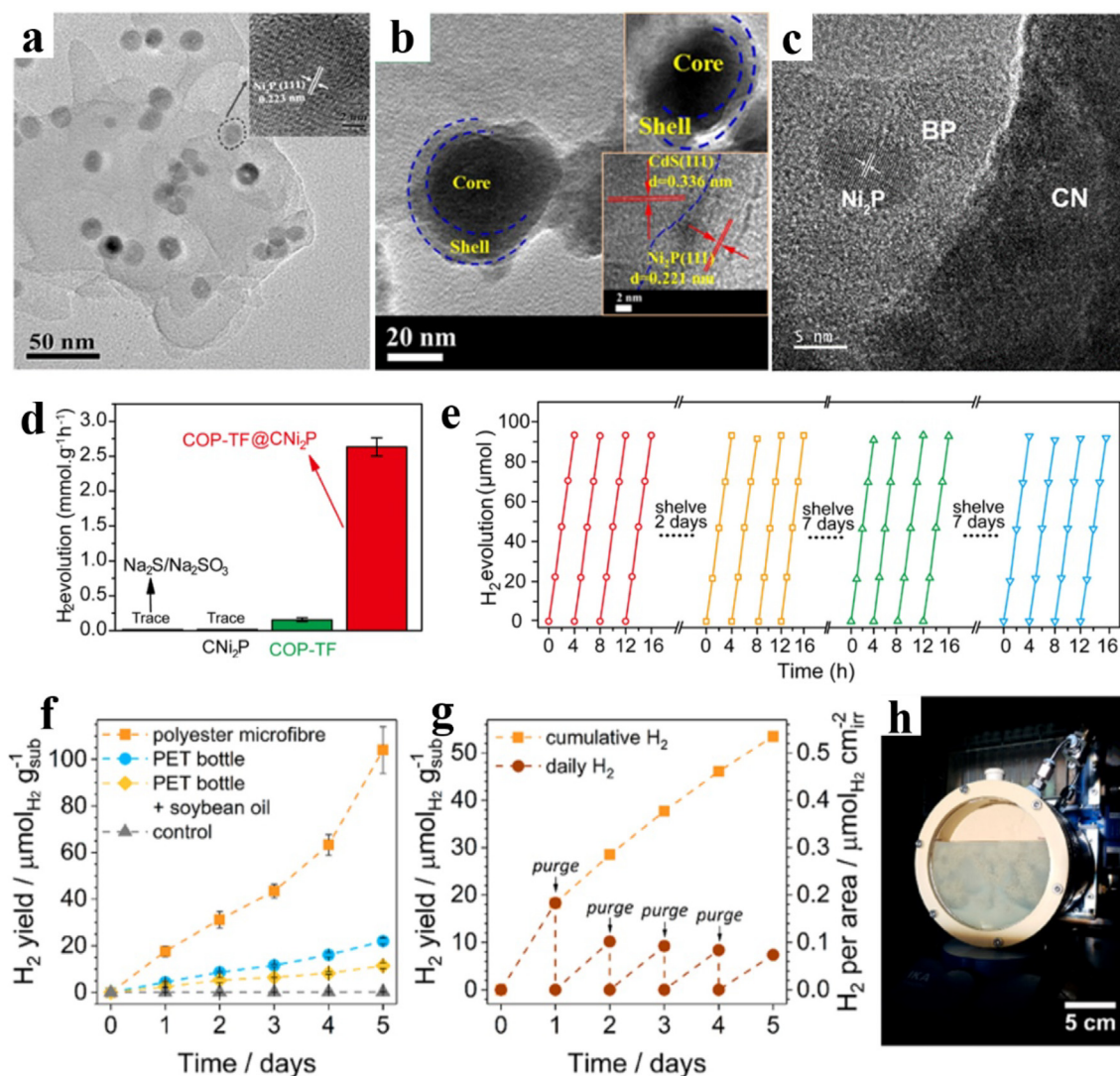
microplastics are widely distributed in oceans, lakes, and drinking water, which will pose a great threat to human health. Photoreforming provides a novel strategy by not only reducing plastic pollution but also producing H<sub>2</sub> and valuable chemical products. For example, Uekert et al. [159] utilized CN<sub>x</sub>/Ni<sub>2</sub>P photocatalyst to produce H<sub>2</sub> and organic chemicals from poly(ethylene terephthalate) (PET) and poly(lactic acid) (PLA) under alkaline conditions. The strong binding of the Ni<sub>2</sub>P cocatalyst to CN<sub>x</sub> improved charge separation, thereby promoting photocatalytic activity. Moreover, CN<sub>x</sub>/Ni<sub>2</sub>P could also effectively and stably reform real-world polymer samples to produce H<sub>2</sub>, as shown in Fig. 10f–h.

Although nickel phosphides cocatalysts can significantly improve the activity of photocatalytic H<sub>2</sub> production, the current researches mainly focus on the phase of Ni<sub>2</sub>P, and there are few studies on other phases of nickel phosphides in application of photocatalytic H<sub>2</sub> production. The nickel-to-phosphorus ratio in nickel phosphides plays an important role in influencing the HER performance. Thus, clarifying the relationship between the phase of nickel phosphides and the activity of photocatalytic H<sub>2</sub> production is the key point for the future studies.

**Table 3**  
Nickel phosphides cocatalysts for photocatalytic H<sub>2</sub> production.

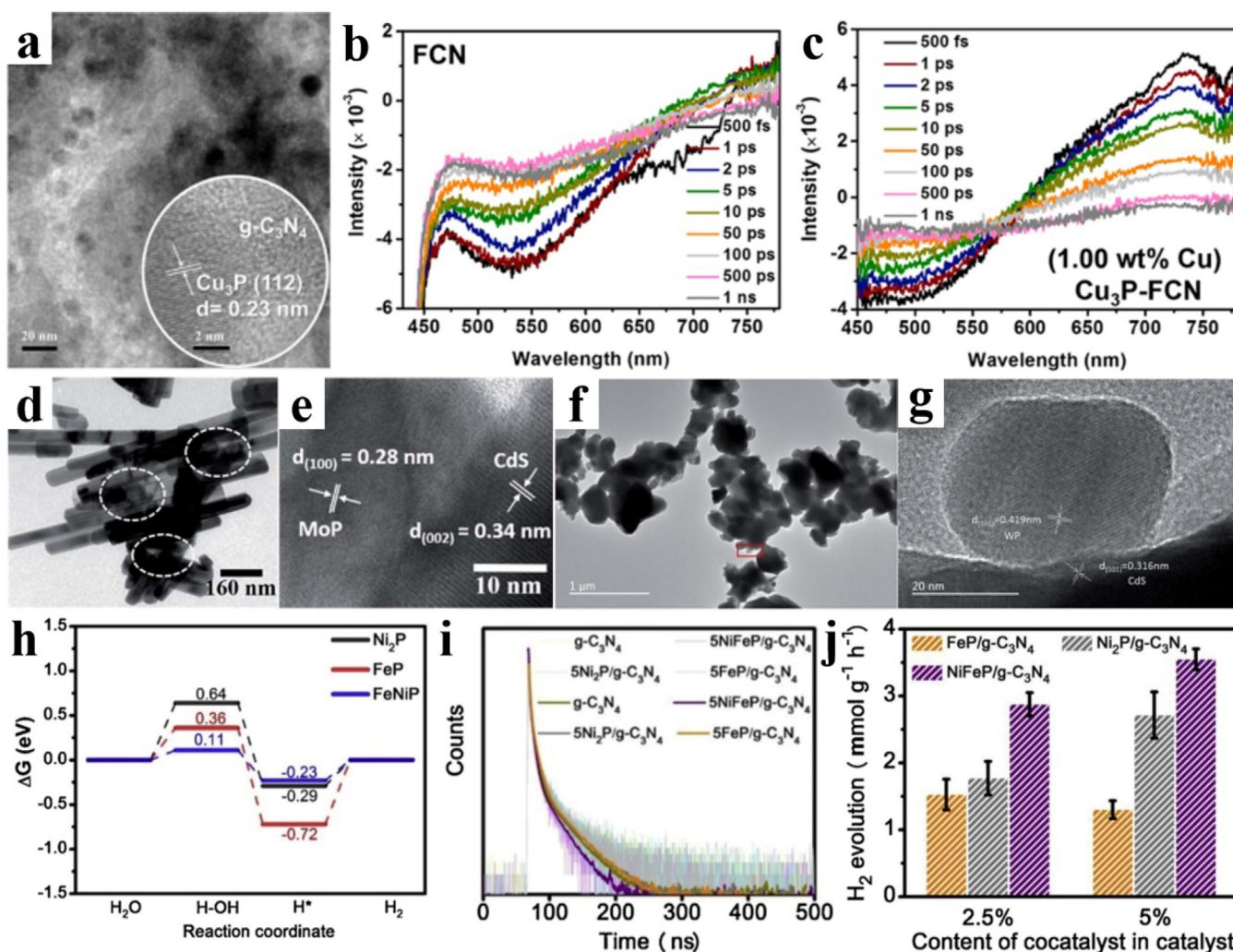
Photocatalyst	Cocatalyst	Light source	Sacrificial agents	Activity( $\mu\text{mol h}^{-1} \text{g}^{-1}$ )	AQY (%) $\lambda$ (nm)	Ref. (year)
CdS/Ni <sub>2</sub> P/g-C <sub>3</sub> N <sub>4</sub>	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	–	15.66	0.18 (420)	[63] (2019)
NiO/Ni <sub>2</sub> P/CN	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	504	0.22 (400)	[145] (2019)
Ni <sub>2</sub> P@BP/CN	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	858.2	2.80 (420)	[146] (2018)
Ni <sub>2</sub> P/g-C <sub>3</sub> N <sub>4</sub>	Ni <sub>2</sub> P	$\lambda > 400 \text{ nm}$ (Xe)	TEOA	2337.09	3.98 (420)	[147] (2019)
Ni <sub>2</sub> P/g-C <sub>3</sub> N <sub>4</sub>	Ni <sub>2</sub> P	UV-Vis (Xe)	TEOA	3344	9.10 (420)	[148] (2018)
C-ZrO <sub>2</sub> /g-C <sub>3</sub> N <sub>4</sub> /Ni <sub>2</sub> P	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	10,040	35.5 (420)	[149] (2018)
Ni <sub>2</sub> P/SNO/CdS-D	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	11,992	35.8 (420)	[150] (2020)
Ni <sub>2</sub> P/CdS/g-C <sub>3</sub> N <sub>4</sub>	Ni <sub>2</sub> P	$\lambda > 400 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	44,450	46.3 (420)	[151] (2017)
Ni <sub>2</sub> P/MIL-125-NH <sub>2</sub>	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	894	27 (400)	[152] (2018)
COP-TF@CNi <sub>2</sub> P	Ni <sub>2</sub> P	$\lambda > 400 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	2500	2.50 (400)	[91] (2019)
Ni <sub>2</sub> P/Cd <sub>0.5</sub> Zn <sub>0.5</sub> S	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	1312	29 (420)	[153] (2018)
Ni <sub>2</sub> P/Zn <sub>0.5</sub> Cd <sub>0.5</sub> S	Ni <sub>2</sub> P	$\lambda > 420 \text{ nm}$ (LED)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	21,190	21.16 (450)	[154] (2018)
Ni <sub>12</sub> P <sub>5</sub> /g-C <sub>3</sub> N <sub>4</sub>	Ni <sub>12</sub> P <sub>5</sub>	$\lambda > 420 \text{ nm}$ (Xe)	TEOA	535.7	4.67 (420)	[143] (2017)
Ni <sub>12</sub> P <sub>5</sub> /ZnIn <sub>2</sub> S <sub>4</sub>	Ni <sub>12</sub> P <sub>5</sub>	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	2263	20.50 (420)	[155] (2019)
CdS@CuS-Ni <sub>x</sub> P	Ni <sub>x</sub> P	$\lambda > 420 \text{ nm}$ (Xe)	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub>	18,160	13.06 (420)	[96] (2019)

AQY: apparent quantum yield; Xe: xenon lamp; TEOA: triethanolamine.



**Fig. 10.** (a) High-magnification TEM image along with HRTEM image (inset) of g-C<sub>3</sub>N<sub>4</sub>/3.5% Ni<sub>2</sub>P. Reproduced with permission from Ref. [82] Copyright 2018 Elsevier. (b) HRTEM image of 10Ni<sub>2</sub>P@CdS. Reproduced with permission from Ref. [68] Copyright 2018 Elsevier. (c) HRTEM image of Ni<sub>2</sub>P@BP/CN. Reproduced with permission from Ref. [146] Copyright 2019 Elsevier. (d) Photocatalytic H<sub>2</sub> production ability of CNi<sub>2</sub>P, COP-TF and COP-TF@CNi<sub>2</sub>P samples; (e) Stability test of COP-TF@CNi<sub>2</sub>P during photocatalytic H<sub>2</sub> production under seawater. Reproduced with permission from Ref. [91] Copyright 2019 American Chemical Society. (f) Long-term photoreforming of polyester microfibers, a PET bottle and an oil-coated PET bottle; (g) Upscaled photoreforming of polyester microfibers; (h) Photograph of the batch reactor in use. Reproduced with permission from Ref. [159] Copyright 2019 American Chemical Society.





**Fig. 11.** (a) TEM and HRTEM (inset) images of  $\text{Cu}_3\text{P}$ -FCN; fs-TA spectra at various delay times of (b) FCN and (c)  $\text{Cu}_3\text{P}$ -FCN. Reproduced with permission from Ref. [93] Copyright 2019 American Chemical Society. (d) TEM and (e) HRTEM images of MoP/CdS nanorods. Reproduced with permission from Ref. [169] Copyright 2015 Royal Society of Chemistry. (f) TEM and (g) HRTEM images of WP/CdS. Reproduced with permission from Ref. [170] Copyright 2017 Royal Society of Chemistry. (h) Free energy diagrams for  $\text{H}_2\text{O}$  reduction to  $\text{H}_2$  by the thermochemical model on  $\text{Ni}_2\text{P}$ ,  $\text{FeP}$  and  $\text{NiFeP}$  surface; (i) Time-resolved fluorescence decay spectra of the samples; (j) Photocatalytic  $\text{H}_2$  production rates of  $\text{FeP/g-C}_3\text{N}_4$ ,  $\text{Ni}_2\text{P/g-C}_3\text{N}_4$  and  $\text{NiFeP/g-C}_3\text{N}_4$ . Reproduced with permission from Ref. [177] Copyright 2019 Elsevier.

#### 5.4. Others

Copper is one of the most abundant elements on the earth. Recently, p-type  $\text{Cu}_3\text{P}$  has been reported to be an ideal cocatalyst for photocatalytic  $\text{H}_2$  production [160]. Sun et al. [161] loaded  $\text{Cu}_3\text{P}$  on CdS nanorods to form a p-n junction for accelerating charge transfer and enhancing photocatalytic  $\text{H}_2$  production activity. The optimized  $\text{Cu}_3\text{P}/\text{CdS}$  presented excellent photocatalytic  $\text{H}_2$  production rate of about  $200000 \mu\text{mol h}^{-1} \text{g}^{-1}$  under visible light irradiation, with the AQY of about 25% at 450 nm excitation. Yue et al. [162] found that the photocatalytic  $\text{H}_2$  production rate of  $\text{Cu}_3\text{P}/\text{TiO}_2$  was 11 times higher than that of pure  $\text{TiO}_2$ , which was attributed to the strong interaction between  $\text{Cu}_3\text{P}$  and  $\text{TiO}_2$  induced accelerated separation of photogenerated electron-hole pairs. Moreover, Shen et al. [163] demonstrated that  $\text{Cu}_3\text{P}$  played an important role in improving the photocatalytic  $\text{H}_2$  production activity of  $\text{g-C}_3\text{N}_4$ . To better understand the  $\text{Cu}_3\text{P}$ -induced charge antitrapping behavior in  $\text{g-C}_3\text{N}_4$  (Fig. 11a), Wang et al. [93] employed femtosecond transient absorption (fs-TA) spectroscopy measurements to reveal the dynamics of charge carriers. As displayed in Fig. 11b-c,  $\text{Cu}_3\text{P/g-C}_3\text{N}_4$  possessed a prolonged photogenerated excited electron lifetime (209 ps), indicating that  $\text{Cu}_3\text{P}$  could effectively promote

transfer of photogenerated charge and inhibit recombination of photogenerated electron-hole pairs. Accordingly, the photocatalytic  $\text{H}_2$  production rate of  $\text{Cu}_3\text{P/g-C}_3\text{N}_4$  ( $277.2 \mu\text{mol h}^{-1} \text{g}^{-1}$ ) was 370 times higher than that of pure  $\text{g-C}_3\text{N}_4$ .

Among the earth abundant TMPs, molybdenum phosphides and tungsten phosphides have been determined as promising cocatalysts for photocatalytic  $\text{H}_2$  production [164–168]. Yue et al. [169] attached MoP nanoparticles on the surface of CdS nanorods to form an intimate attachment (Fig. 11d and e) for  $\text{H}_2$  production under visible light irradiation. The  $\text{H}_2$  production rate of the optimized MoP/CdS ( $163200 \mu\text{mol h}^{-1} \text{g}^{-1}$ ) was  $> 20$  times higher than that of the pure CdS, which was ascribed to the suitable Fermi level alignment induced fast charge transfer in the interface of MoP and CdS. Meanwhile, Zhang et al. [170] found that WP could be used as an effective cocatalyst for improving the photocatalytic  $\text{H}_2$  production activity of CdS. As exhibited in Fig. 11f and g, they deposited WP on the CdS support by a ball-milling method to generate a compact solid-solid interface. Benefiting from the suppressed recombination of the photogenerated electron-hole pairs, the WP/CdS obtained a superior  $\text{H}_2$  production rate of  $3104 \mu\text{mol h}^{-1} \text{g}^{-1}$ , which was 11.67 times higher than that of the pure CdS.



In comparison with monometallic phosphides, bimetallic phosphides have a higher activity due to the synergistic effect of metallic elements [171–173]. Both of the active components in bimetallic phosphides can act as acceptors and transporters of electrons to facilitate the separation and transfer of charges, and further promote photocatalytic  $H_2$  production [70,174,175]. For example, Xue et al. [176] decorated CoNiP onto PCN nanosheets for highly efficient photocatalytic  $H_2$  production. Compared to  $Co_2P$  and  $Ni_2P$ , CoNiP was more conducive to the formation of H-H bond and the desorption of  $H_2$  molecule. And it was found that the formation of  $P^+-P^{\delta-}Co/Ni^{\delta+}$  chemical bridge between CoNiP and PCN could significantly boost charge transfer and separation. Thus, the photocatalytic  $H_2$  production rate of CoNiP-PCN was up to  $239.3 \mu mol h^{-1} g^{-1}$ , which was higher than that of the  $Co_2P$ -PCN and  $Ni_2P$ -PCN. Zhu et al. [177] found that the absolute value of Gibbs free energies of NiFeP were reduced compared with  $Ni_2P$  and  $Fe_2P$  (Fig. 11h), which would be beneficial for the dissociation of  $H_2O$  and evolution of  $H_2$ . After introducing NiFeP into g- $C_3N_4$ , the fluorescent lifetime was decayed from 8.77 ns to 5.21 ns, as shown in Fig. 11i. This result demonstrated that the photogenerated electrons of g- $C_3N_4$  were directly transferred to the surface of NiFeP, resulting in the worse delocalization ability of electrons and its easier reactions with protons to produce  $H_2$ . Therefore, the NiFeP/g- $C_3N_4$  possessed a higher  $H_2$  production rate ( $3549 \mu mol h^{-1} g^{-1}$ ) than  $Ni_2P/g-C_3N_4$  FeP/g- $C_3N_4$  (Fig. 11j).

## 6. Conclusions

In conclusion, this review summarizes the recent developments of TMPs cocatalyst in photocatalytic  $H_2$  production. Firstly, the functions of TMPs in photocatalytic  $H_2$  production are presented, mainly including enhancing light absorption, providing active sites, accelerating charge transfer, lowering overpotential and strengthening photostability. Then, the synthetic strategies of TMPs such as solid-phase synthetic method and solution-phase synthetic method are introduced. Meanwhile, the loading methods of TMPs on semiconductor photocatalysts are summarized, such as post loading method and in situ reduction method. After that, the applications and mechanisms of the common TMPs (iron phosphides, cobalt phosphides, nickel phosphides, copper phosphides, molybdenum phosphides, tungsten phosphides and bimetallic phosphides) in photocatalytic  $H_2$  production are discussed in detail.

## 7. Perspectives

Although some achievements in TMPs have been made, there are still many challenges to utilize TMPs cocatalysts for significant improvement in photocatalytic  $H_2$  production. Some points are listed as follows.

- (1) Safer and quicker strategies for synthesizing TMPs should be developed. Most synthetic strategies to prepare TMPs require long reaction times, high reaction temperature and even protection of an inert atmosphere. Meanwhile, the toxic tail gas and the remaining organic solvent increase the environmental risks. Therefore, it is urgently to develop novel synthetic strategies for safely and quickly preparing TMPs.
- (2) Loading methods of TMPs on semiconductors should be optimized. Because of the low energy barrier at the interface of intimate junction, it is critical to optimize the loading methods for the fabrication of atomically well-bonded junction, which will significantly accelerate the charge transfer. In addition, such an intimate junction between the TMPs cocatalyst and semiconductors will strengthen photostability of the composite, which is beneficial for the practical application.
- (3) New types of TMPs cocatalysts and semiconductors should be explored. Among the various TMPs, iron phosphides, cobalt

phosphides and nickel phosphides are the most frequently used cocatalysts in photocatalytic  $H_2$  production. Although many other kinds of TMPs have proven to be efficient electrocatalysts for HER, the investigation on their application in photocatalytic  $H_2$  application is still insufficient. Thus, it is very important to develop other types of TMPs for photocatalytic  $H_2$  production. Meanwhile, in addition to combining the TMPs with the commonly used  $TiO_2$ , CdS and g- $C_3N_4$ , other novel semiconductors should also be considered.

- (4) Highly active TMPs with different morphologies and structures should be developed. Currently, the morphology of most TMPs used for photocatalytic  $H_2$  production is nanoparticles, which are easy to self-aggregate, resulting in a reduction in the number of active sites in photocatalytic reaction. Therefore, it is urgent to develop nanostructured TMPs with abundant active sites for assisting photocatalytic  $H_2$  production, such as 2D ultrathin nanosheets, 0D quantum dots and single atoms.
- (5) Properties and functional mechanisms of the loaded photocatalysts should be deeply investigated. The effects of TMPs on the photocatalytic activity of the composites have been extensively studied. Nevertheless, the fundamental investigations on the physicochemical properties, photogenerated charge carrier dynamics and  $H_2$  evolution mechanisms of loaded photocatalysts are scarce. Therefore, it is necessary to use some advanced technologies to solve these issues, such as in situ X-ray absorption spectroscopy and femtosecond transient absorption spectroscopy. Moreover, DFT calculations can be utilized to better design experiments and clarify mechanisms.
- (6) Impacts of sacrificial agents should be concerned. In photocatalytic  $H_2$  production, sacrificial agents are usually added into the system to consume photogenerated holes for inhibiting the recombination of photogenerated electron-hole pairs. However, the addition of sacrificial agents may cause pollution to water, limiting sustainable practical applications. Combining  $H_2$  evolution half-reaction with organic contaminants degradation or selective oxidation can resolve this issue. Accordingly, efforts should be made to replace sacrificial agents with the organic contaminants or the precursors of valuable chemical products in the future work.
- (7) Overall water splitting should be considered. Current researches on TMPs cocatalysts are still focused on the  $H_2$  evolution half-reaction. The ultimate objective in developing high efficiency TMPs cocatalysts is to realize the overall water splitting only by utilizing solar energy. Therefore, future studies of TMPs cocatalysts should be focused on the applications on overall water splitting.

## Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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